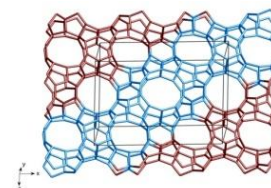
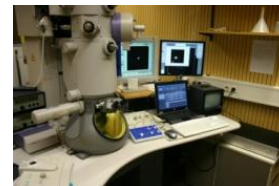
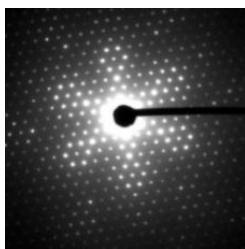
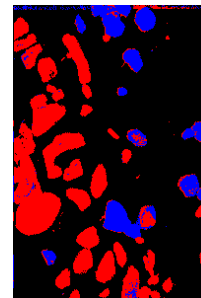
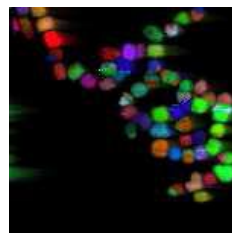
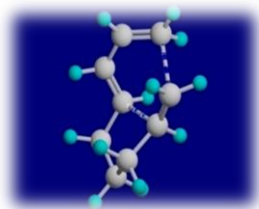


# NEW techniques for TEM nanoanalysis

precession electron diffraction for organic – inorganic nanostructures



**Dr. Stavros Nicolopoulos**  
Consultant IUCr Electron Crystallography Commission

# This document was presented at PPXRD - Pharmaceutical Powder X-ray Diffraction Symposium

Sponsored by The International Centre for Diffraction Data

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PPXRD Website – [www.icdd.com/ppxrd](http://www.icdd.com/ppxrd)

ICDD Website - [www.icdd.com](http://www.icdd.com)

# X-ray Diffraction (single crystal ): all info in reciprocal space

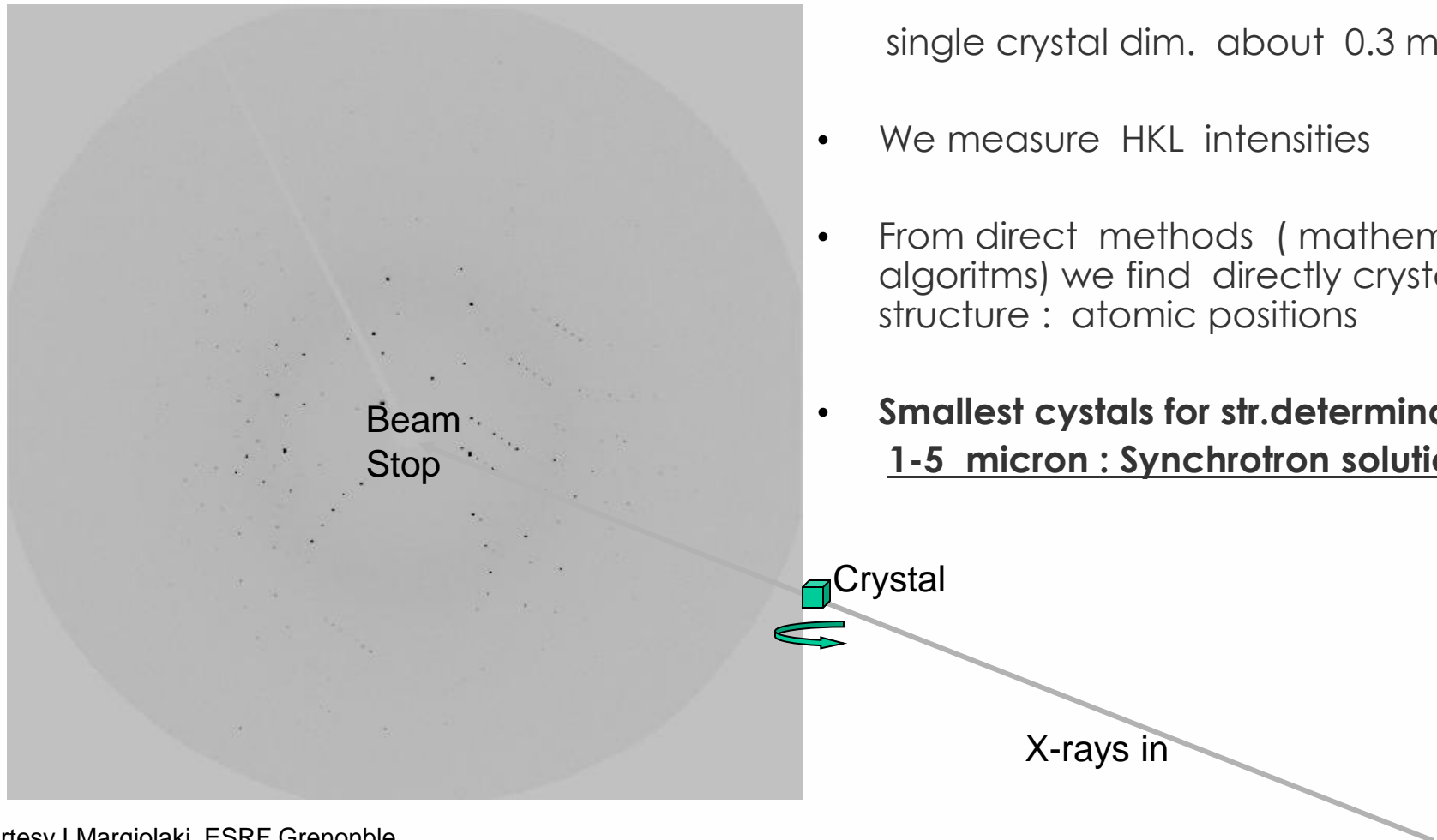
Bragg law

$$n\lambda = 2d\sin\theta$$

- Single X-Ray diffractometer collects set of 3d HKL reflections

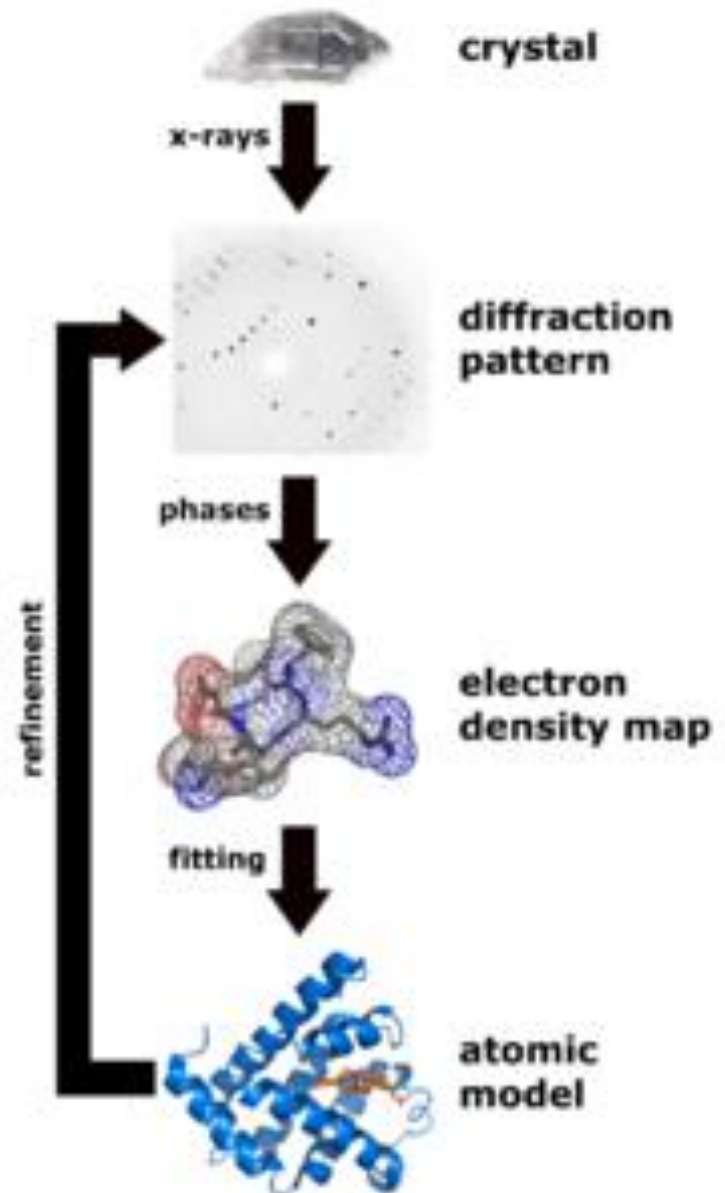
single crystal dim. about 0.3 mm

- We measure HKL intensities
- From direct methods (mathem. algorithms) we find directly crystal structure : atomic positions
- **Smallest crystals for str.determination**  
**1-5 micron : Synchrotron solution**

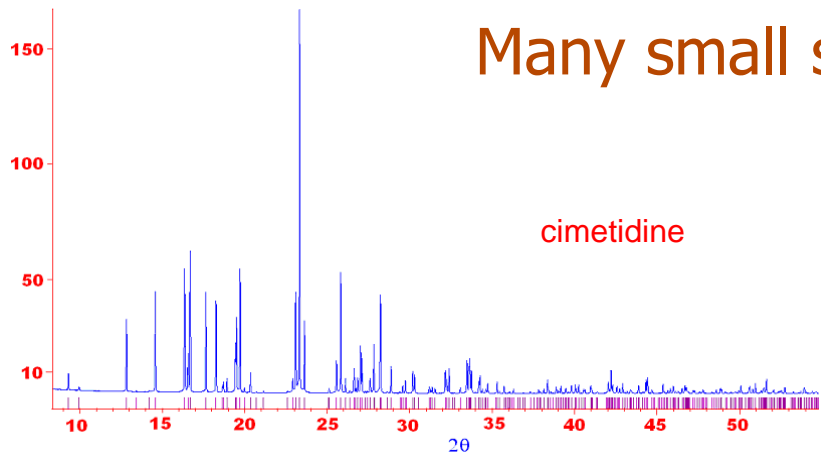


# X-ray Crystallography

X-ray crystallography is the science of determining the arrangement of atoms within a crystal from the manner in which a beam of X-rays is scattered from the electrons within the crystal. The method produces a three-dimensional picture of the density of electrons within the crystal, from which the mean atomic positions, their chemical bonds, their disorder and sundry other information can be derived.

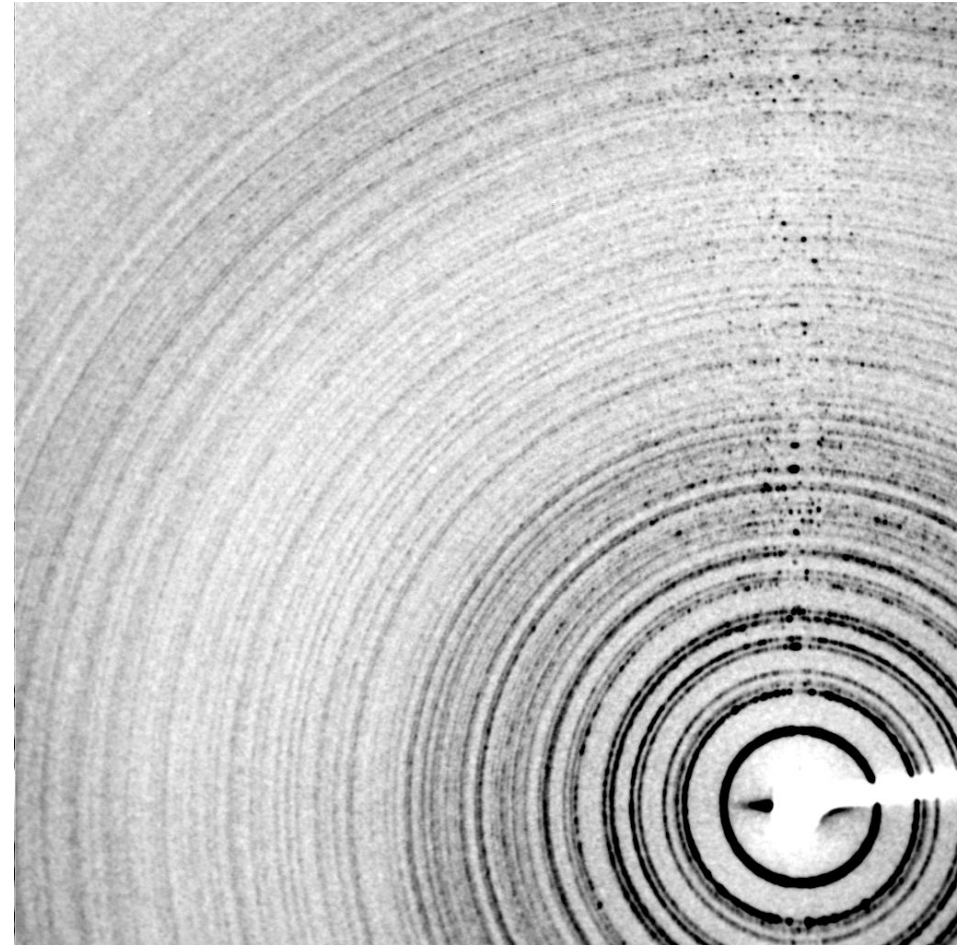


# Many small single crystals make a powder



- » 1
- » 2
- » 3
- » 5
- » 10
- » 20
- » 50

- Spots cover spheres in 3D reciprocal space
- 2D area detector takes a slice
- (on Ewald sphere)
- 1D powder scan measures distance from origin



# X-Ray powder diffraction: limitations

Ovelapping of reflections

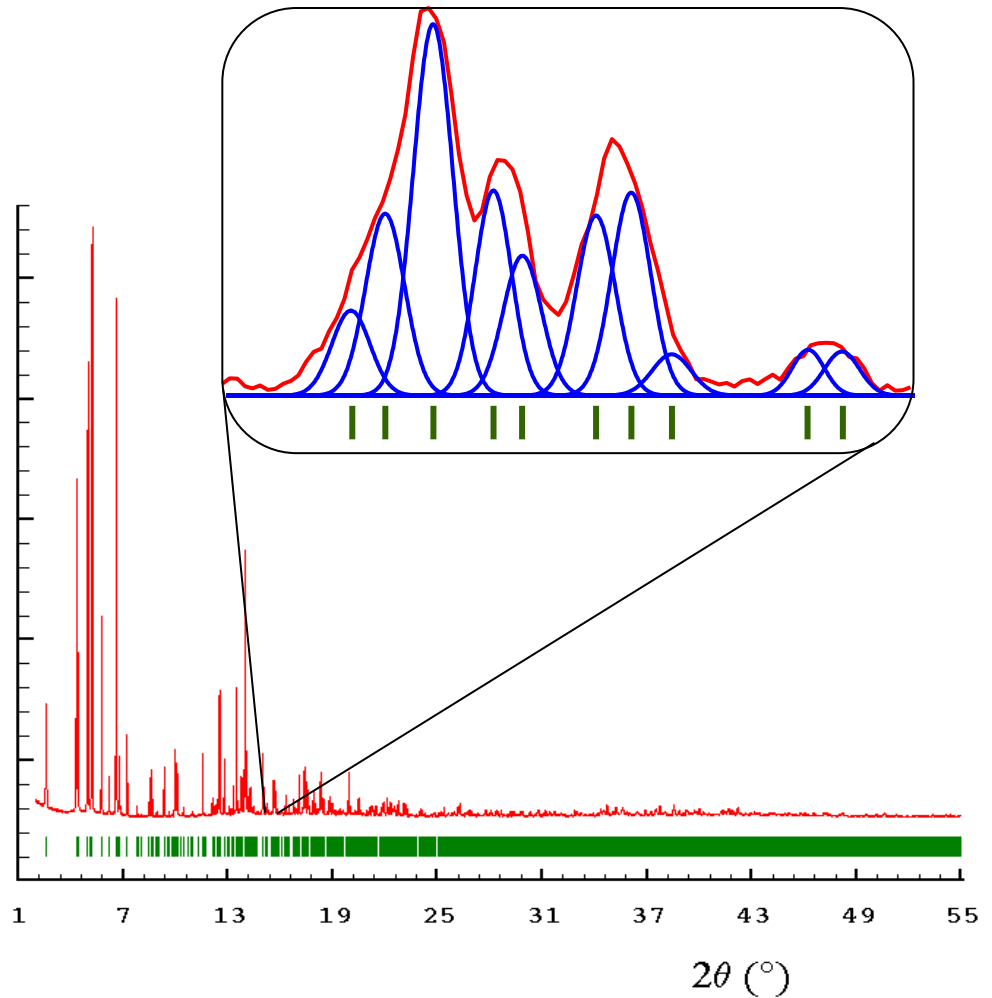
stacking intergrowth

Preferential orientation

Nanocrystals

Impurities

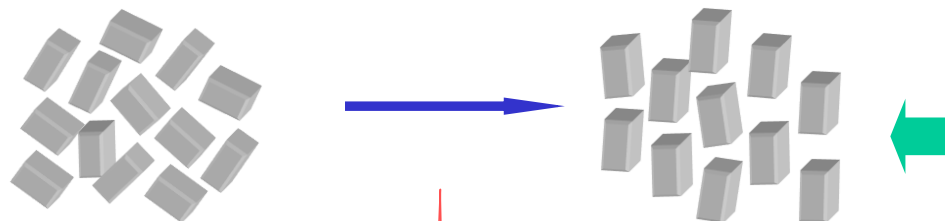
Light atoms (Li, Be)



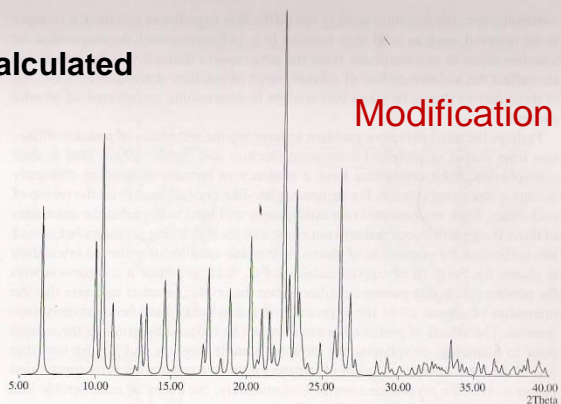
# X- Ray powder diffraction limitations

overlapping

Preferential orientation



calculated



Modification III sulphathiazole

Exper. pattern

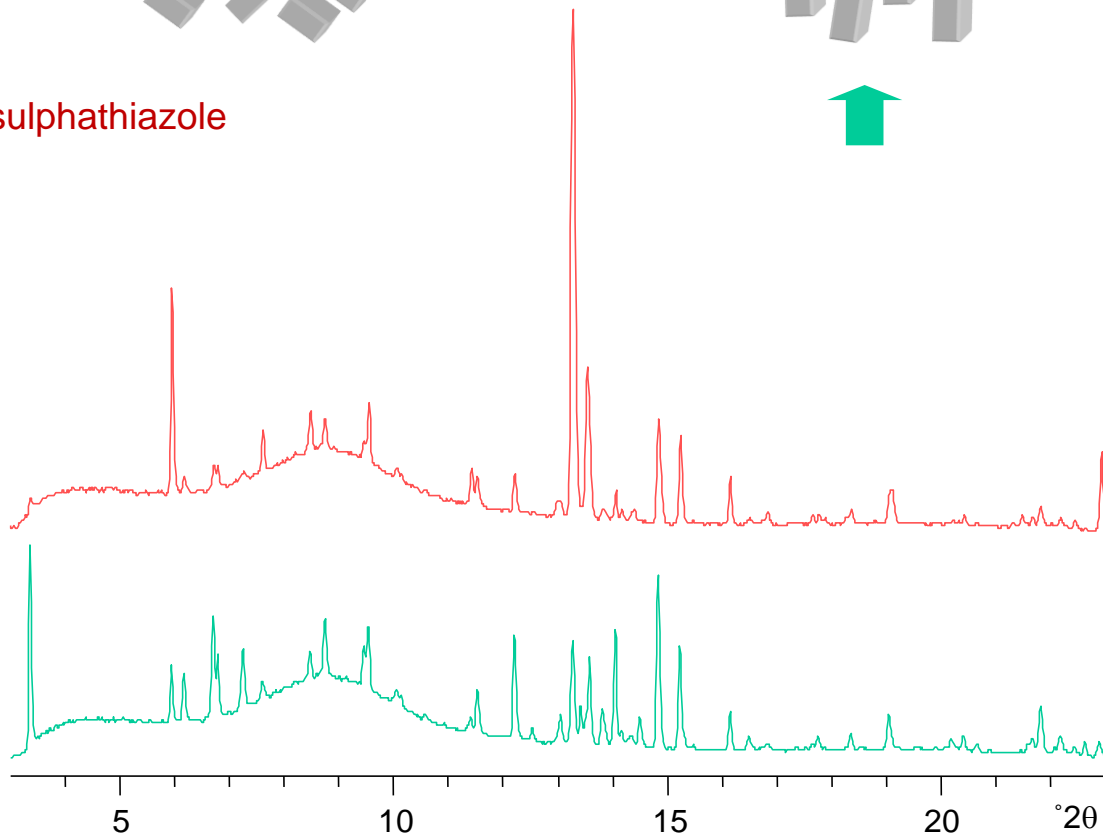
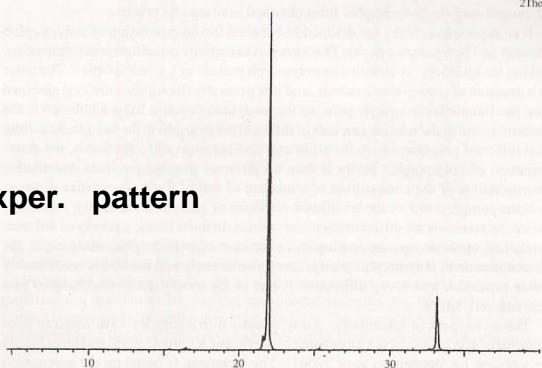


Fig. 4.21 The influence of preferred orientation on the experimental X-ray powder diffraction pattern of modification III of sulphathiazole. Upper, expected pattern calculated from the single crystal structure; lower, experimental powder pattern. (Adapted from Threlfall 1999, with permission.)

From book J.Bernstein Polym in molecular crystals

# X- Ray powder diffraction limitations

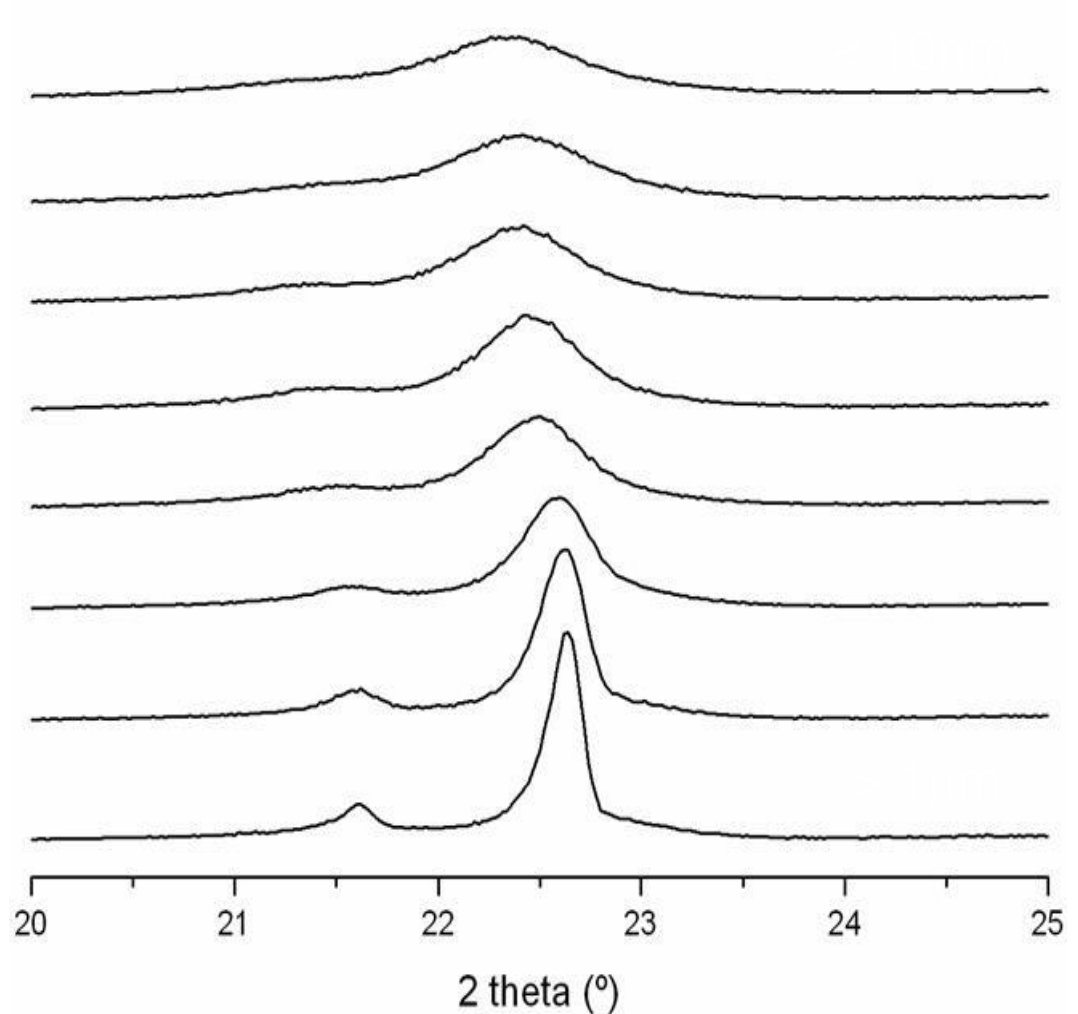
overlapping

Stacking intergrowth

Preferential orientation

Impurities

Light atoms (Li, Be)

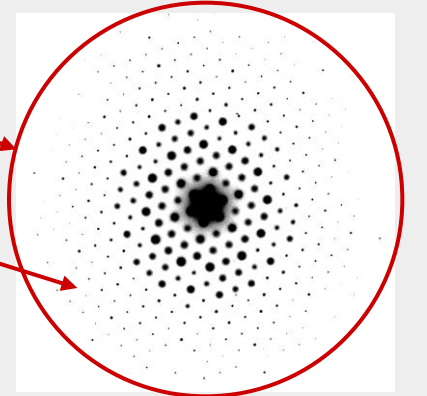
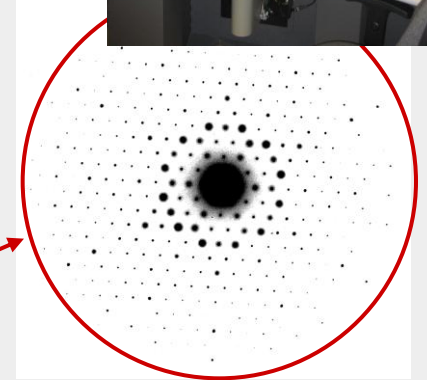
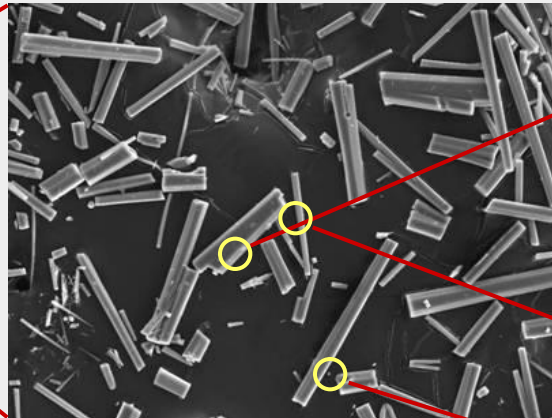




# Electron diffraction in transmission electron microscope (TEM)

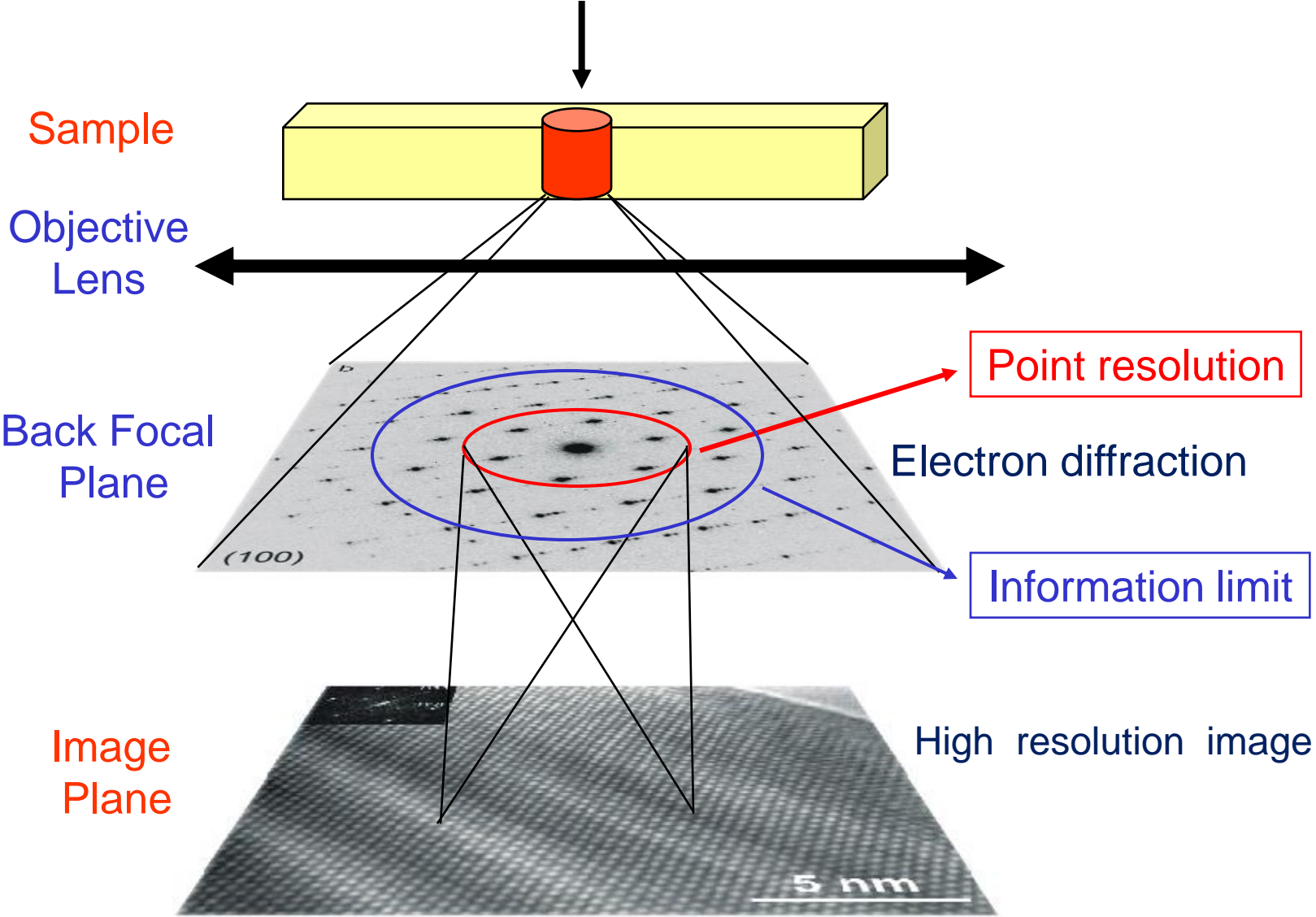


**Powder patterns : sum of individual nanocrystals**



**Cell parameter and symmetry determination from nm single crystals**

# Image formation in TEM



Neutrons, single crystal

X-rays, single crystal  
(laboratory diffractometer)

X-rays, single crystal  
(synchrotron)

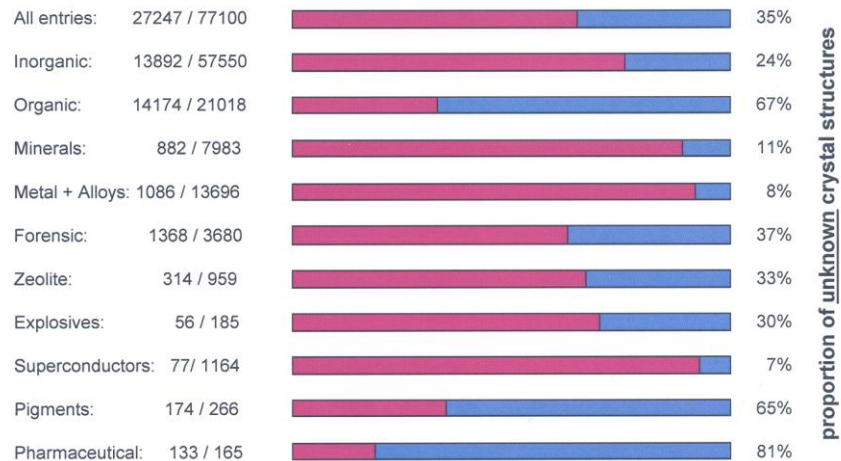
X-rays and neutrons, single phase powder  
(laboratory diffractometer)

X-rays and neutrons, single phase powder  
(synchrotron)

single crystal

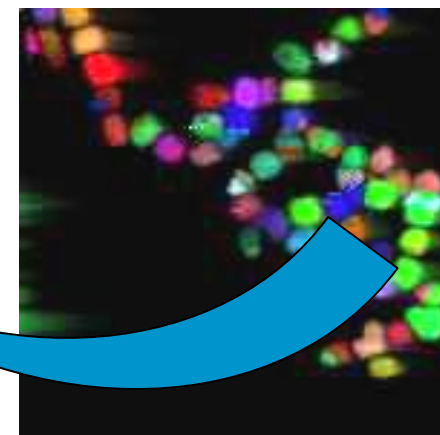
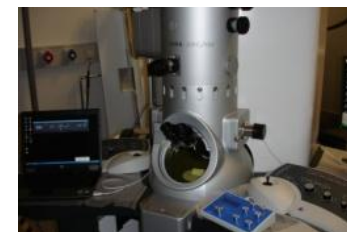
10<sup>6</sup> 10<sup>5</sup> 10<sup>4</sup> 10<sup>3</sup> 10<sup>2</sup> 10<sup>1</sup> 1 nm

3D data 1D data



proportion of unknown crystal structures

Fractions of materials with unknown crystal structures. Data extracted from JCPDS-ICDD Database 1997



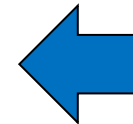
With TEM we can analyze nm size crystals

# Characterization

for new unknown compound



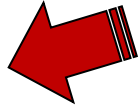
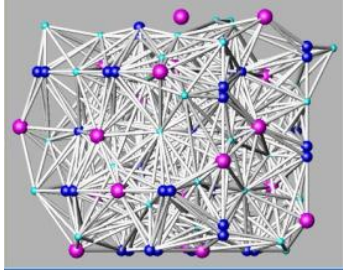
**Find crystal cell parameters**



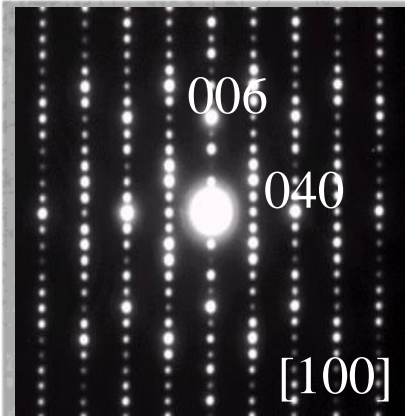
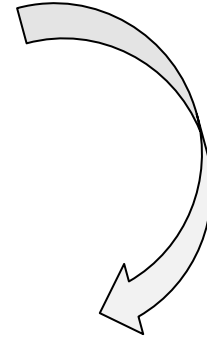
**Find atomic crystal structure**

**Texture – (multi) phase analysis**

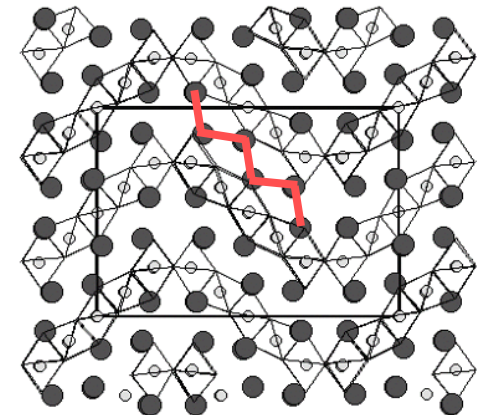
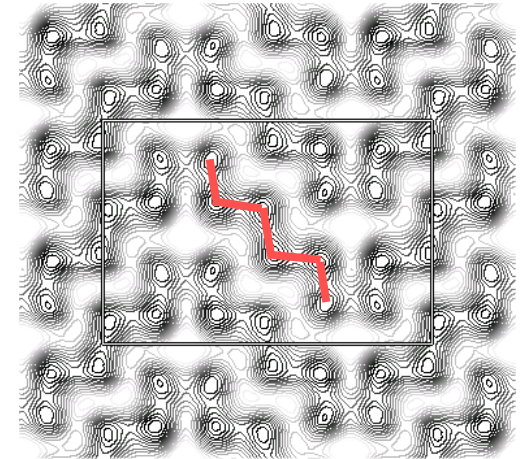
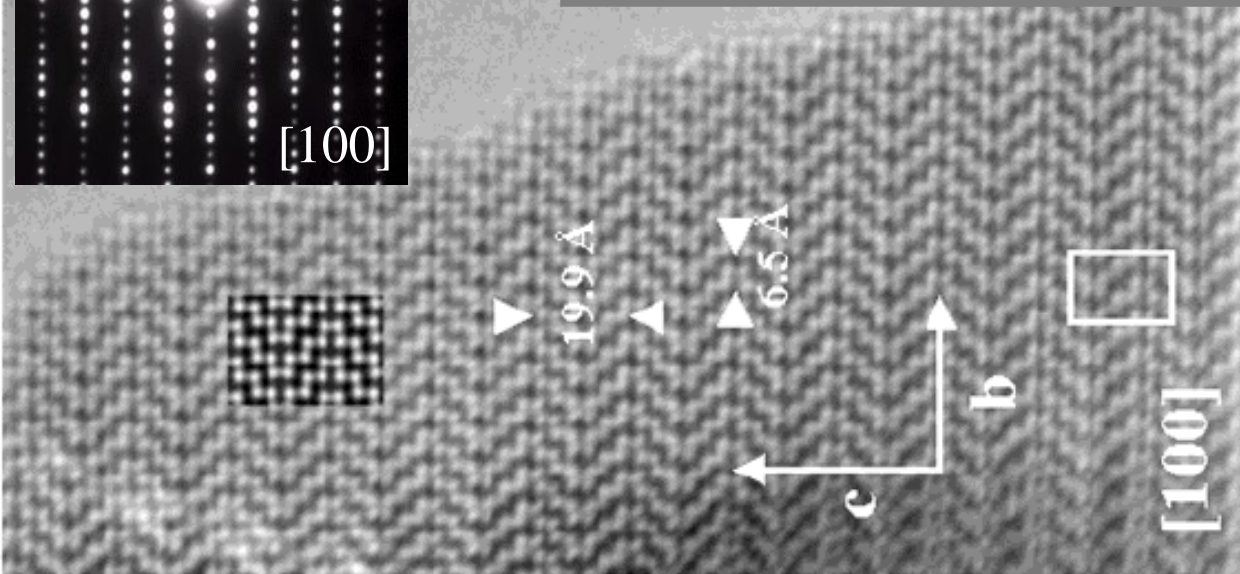
# High resolution TEM : only 2D projected atomic structure



**NOT possible to get  
3D atomic model**

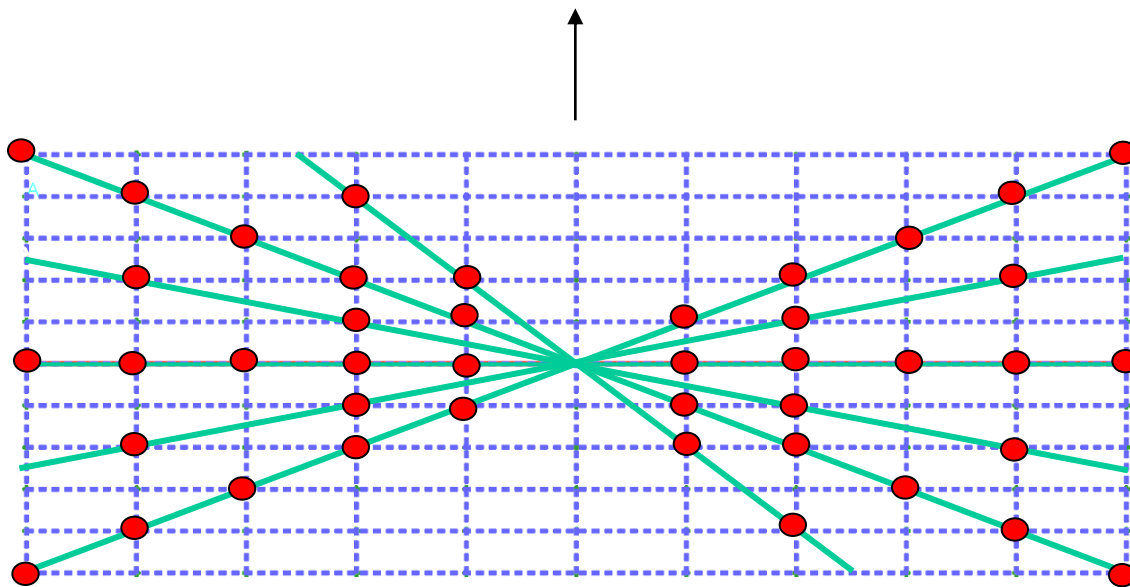
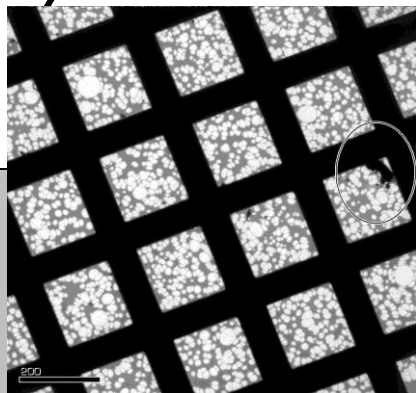
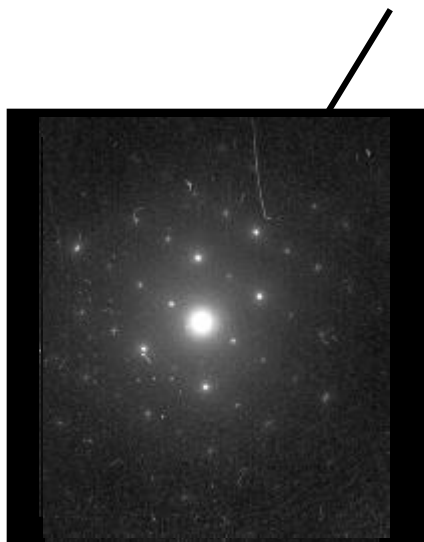
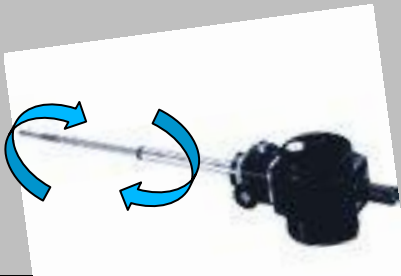


$a=5.70 \text{ \AA}$ ,  $b=13.18 \text{ \AA}$ ,  
 $c=19.92 \text{ \AA}$   
S.G. Cmca(64)

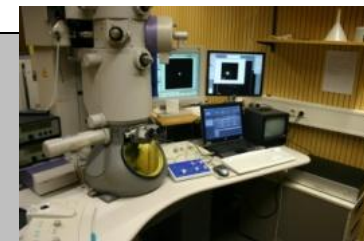


# Find crystal cell parameters

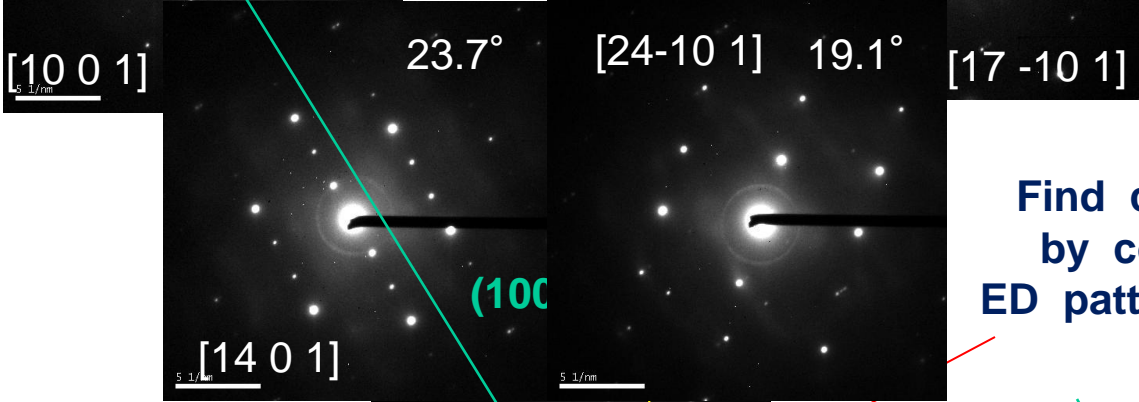
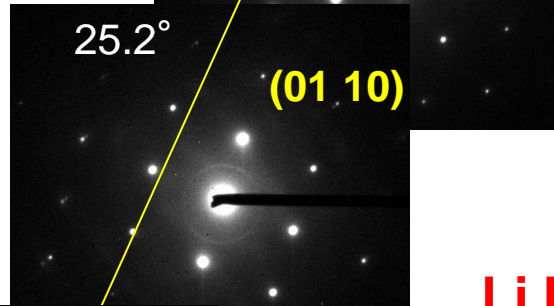
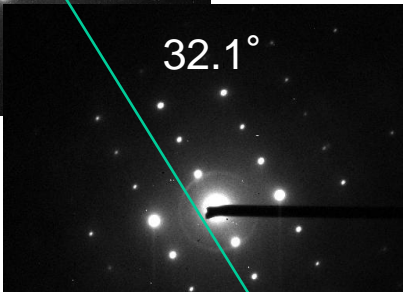
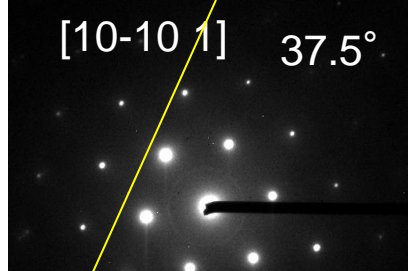
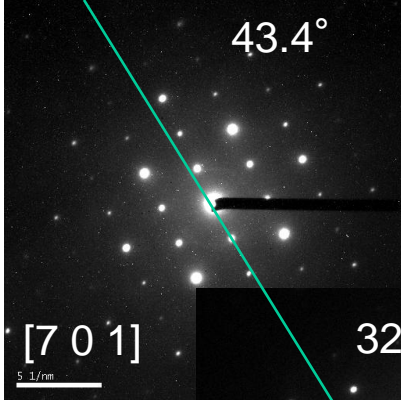
TEM : manual tilt series acquisition



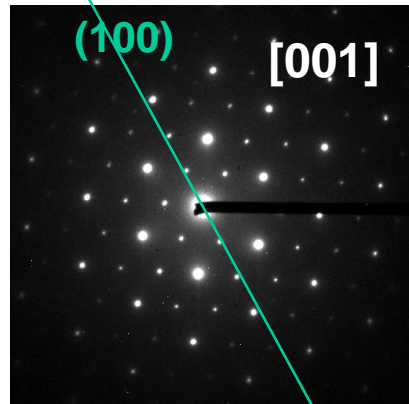
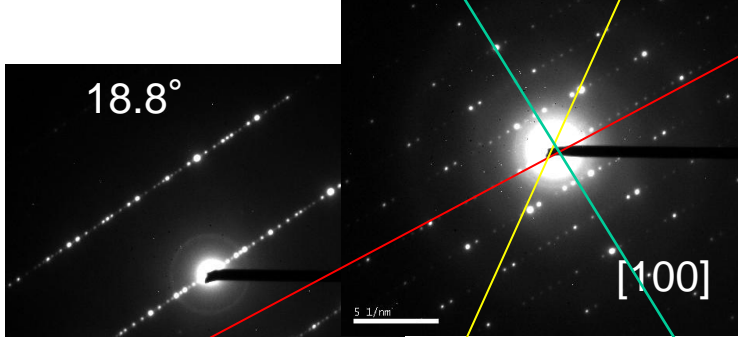
Problem : “missing cone” = lost data



Hexagonal cell  
 $a=5.05 \text{ \AA}$   $c=32.5^\circ$



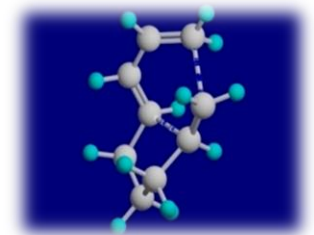
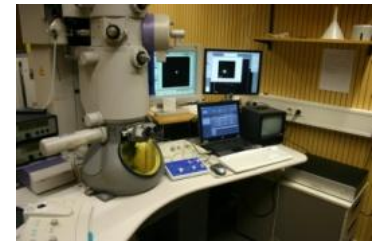
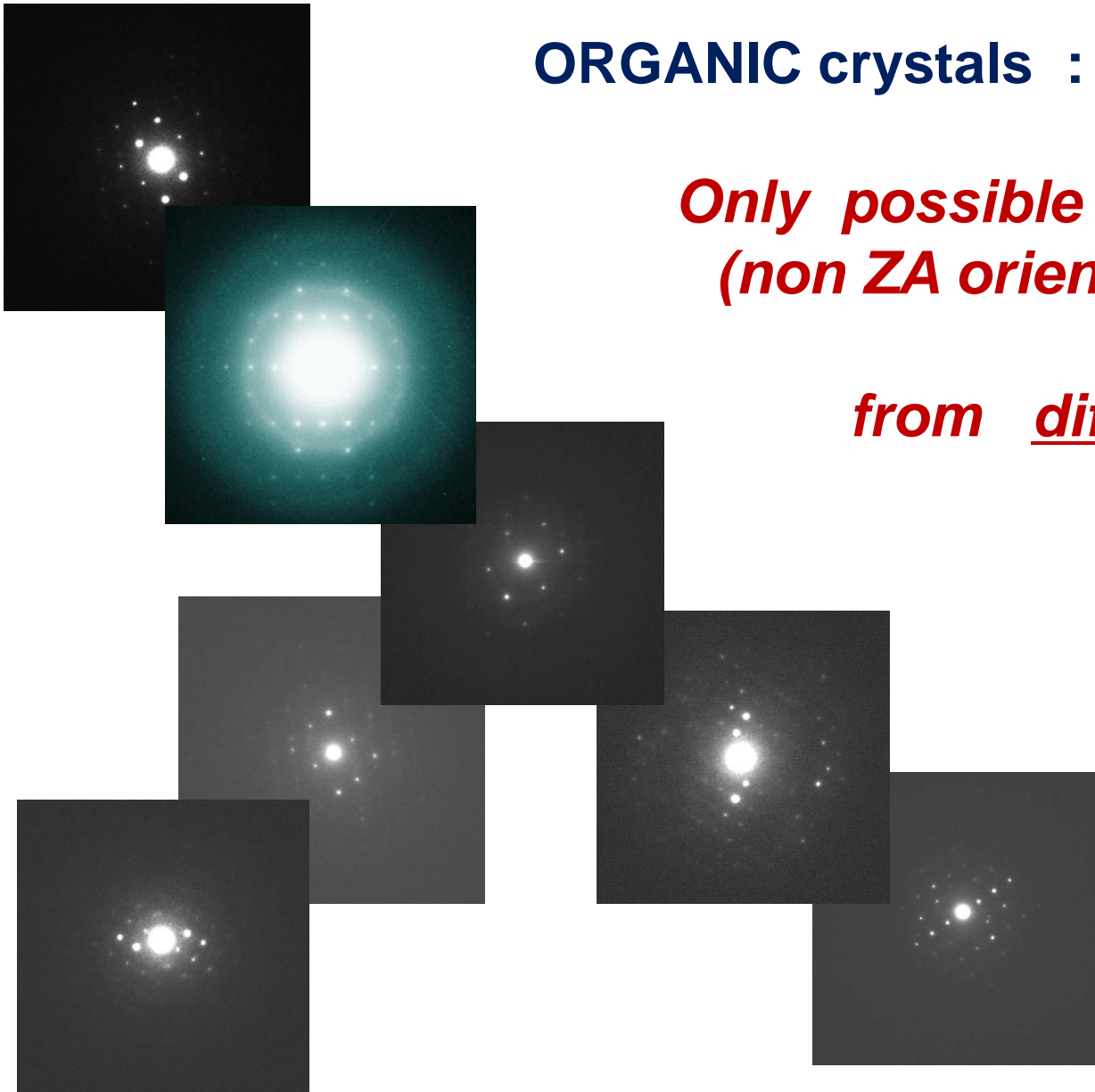
Find crystal cell parameters  
by combining several ZA  
ED patterns from same crystal



# ORGANIC crystals : beam sensitive

*Only possible to collect several  
(non ZA oriented ED patterns)*

*from different crystals*



Penicillin G , courtesy Dr.D.Gueorguieva Leiden Univ NL



# ORGANIC CRYSTALS

Find and index ANY 3d CELL from  
RANDOM diffraction patterns from same or  
several crystals

The image displays a workflow for indexing organic crystal diffraction patterns using the EDiff software suite. The process is shown through several overlapping windows:

- EDiff (Top Left):** The main control window with various parameters for data collection and processing. It includes fields for 'ResolutionRange', 'CrystalSystem' (Cubic, Monoclinic, Hexagonal, Tetragonal, Rhombo), and 'UnitCellParameters'. A search range is set from 5 to 100.
- ElectronDiffractionCheck (Middle Left):** A window showing a diffraction pattern with a grid of blue dots representing indexed spots. It includes a 'Fitting Threshold' and 'Find Neighbors' options.
- Pattern Fitting with Auto-Correlation Image (Right):** A window showing a diffraction pattern with a grid of blue dots and a list of parameters on the right, including 'Unit Cell' and 'Resolution'.
- Indexing Refinement (Bottom Center):** A window showing a diffraction pattern with a grid of blue dots and a list of parameters on the left, including 'Fitting Threshold' and 'Find Neighbors'.

The background features a photograph of a dark, irregularly shaped organic crystal on a light surface, with a scale bar in the bottom right corner.

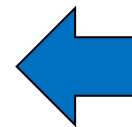
# Characterization

for new unknown compound



**Find crystal cell parameters**

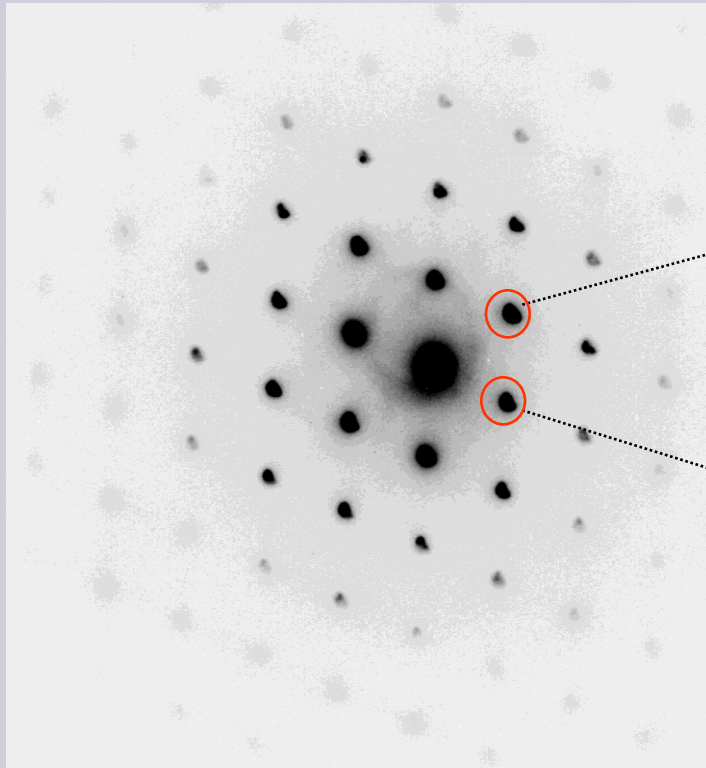
**Find atomic crystal structure**



**Texture – (multi) phase analysis**

# Electron diffraction

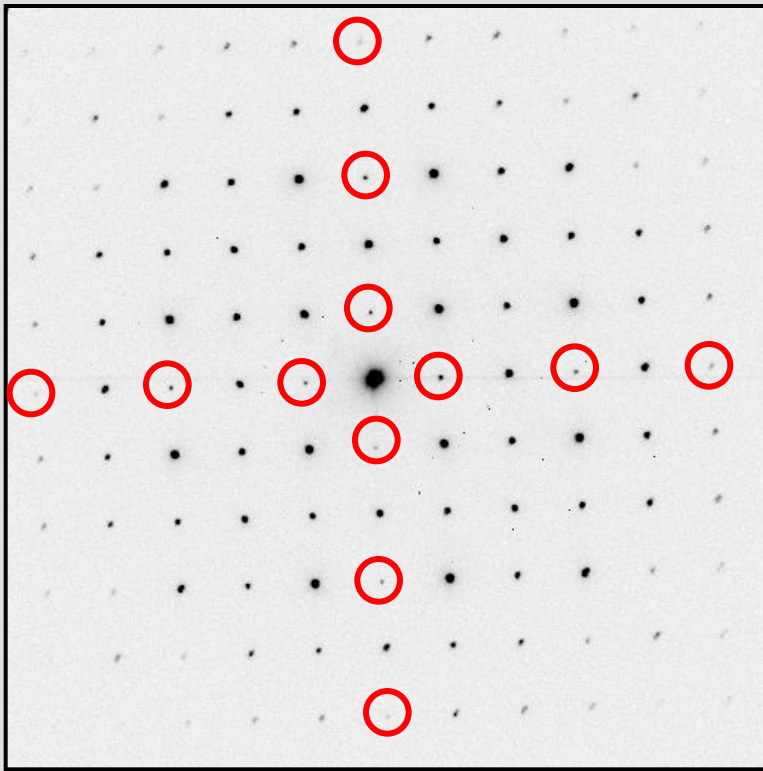
- Electron diffraction is highly dynamical



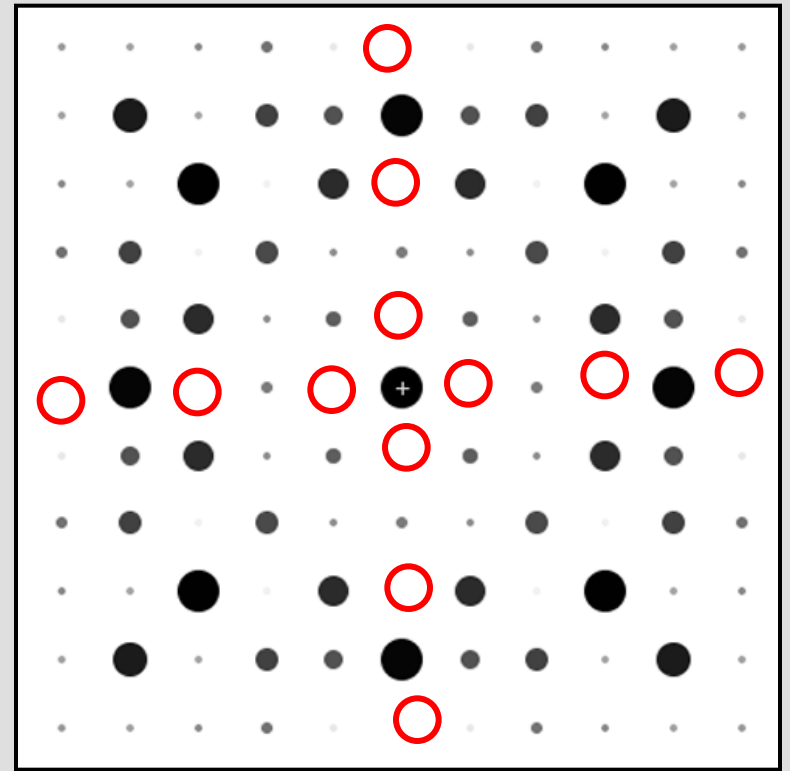
Si [011] spg Fd3m  
*(h00) allowed only if  $h=4n$*

$$I_{\text{exp}} \approx |F(\vec{h})|^2$$

The electron diffraction intensities cannot be used for solving or refining the structure in a straightforward way as X-ray data



Normal electron diffraction pattern  
(dynamical) thickness > 10 nm

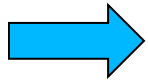


Ideal kinematic diffraction pattern  
( like X-Ray )

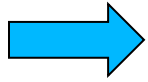
Melilite: tetragonal  $a=b=7.8\text{\AA}$   $c=5.0\text{\AA}$   
 $P-42_1m$

Space group extinction rules :  $(h00) \text{ e } (0k0) \text{ h}=2n, \text{ k}=2n$

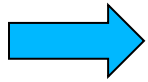
## ***Electron (precession) diffraction : the right way to solve nanostructures in TEM***



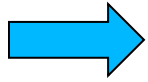
**Electron diffraction intensities ( usually dynamical ) cannot be used ( like X-Ray ) to solve structures, as they lead in wrong structural models**



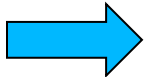
**Precession electron diffraction in TEM has been discovered in Bristol , UK by Vincent and Midgley (1994 , Ultramicroscopy 53, 271)**



**During beam precession , the beam is tilted and precessed at high frequency though the optical axis on a conical surface : crystal is not moving, but Ewald sphere is precessing around the optical axis**



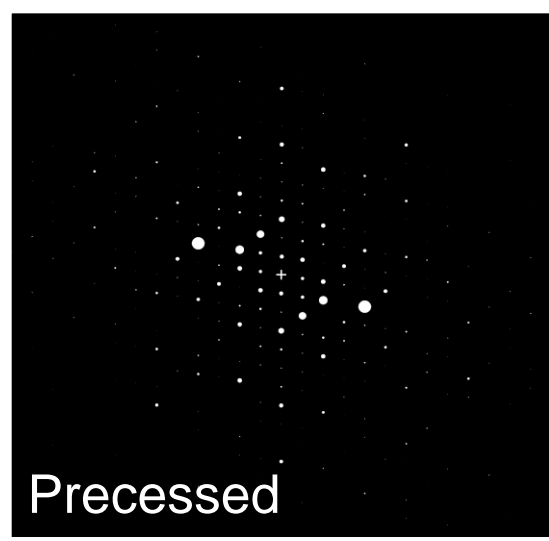
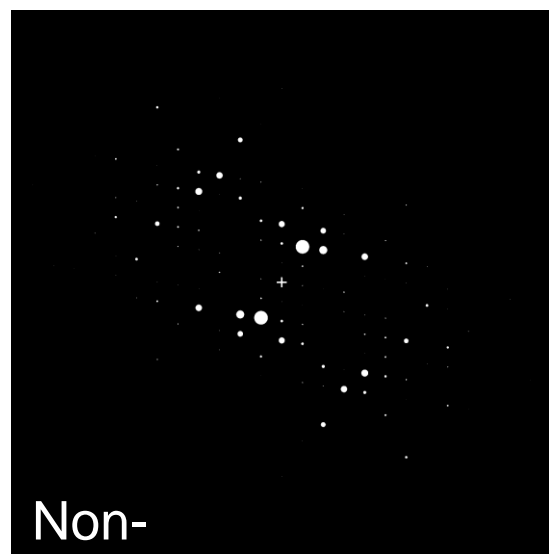
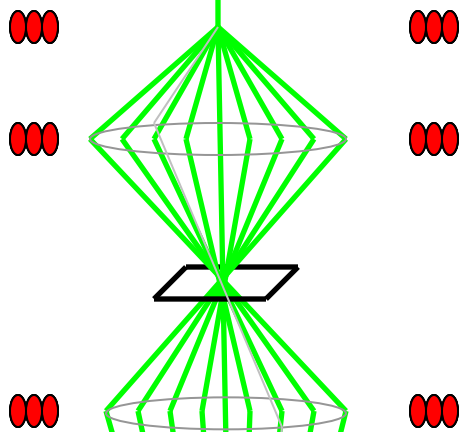
**Due to precession many HKL intensities far out in reciprocal space appear with intensities integrated over the excitation error**



**Precession intensities behave much closer to ideal ( kinematical ) intensities, thereore can be used to solve crystal nanostructures ( Ultramicroscopy, vol.107, issue 6-7, July 2007)**

Scan lens

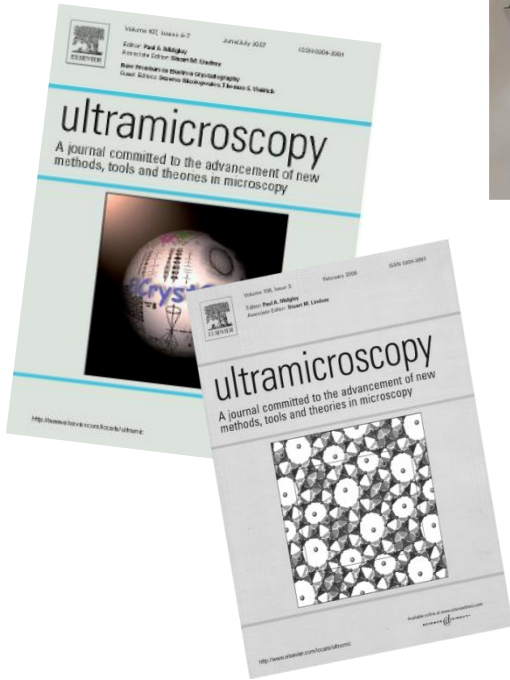
De-scan lens



Precession...

(Diffracted amplitudes)

# DigiSTAR digital precession unit

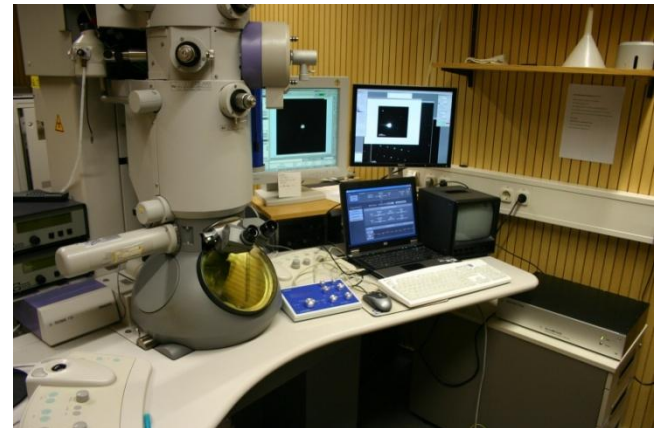
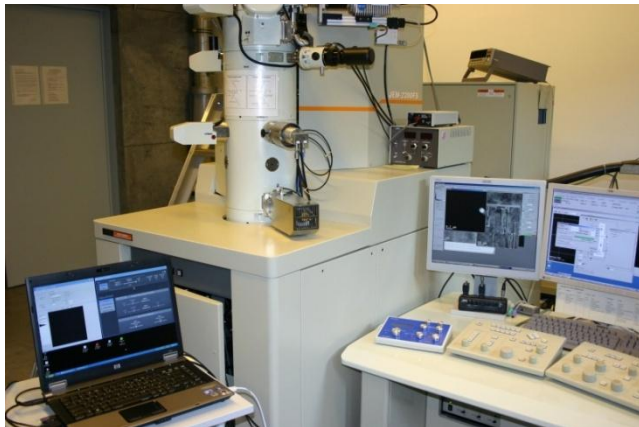
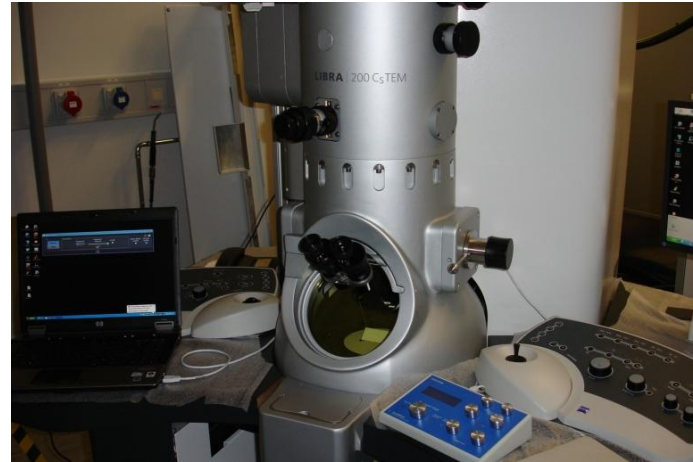
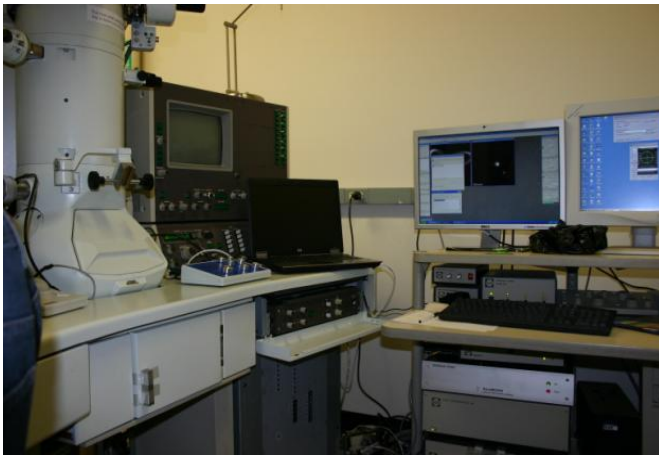


**PRECESSION : DISCOVERED IN UK 1996 R.Vincent- P. Midgley**

**NanoMEGAS : FIRST TEM commercial device**

- **MORE THAN 50 ARTICLES SINCE 2004**
- > **MORE THAN 60 INSTALLATIONS WORLD-WIDE**

# Digital precession interface for advanced TEM



**Zeiss Libra 200F Cs corrected**

**Jeol 2200 FS**

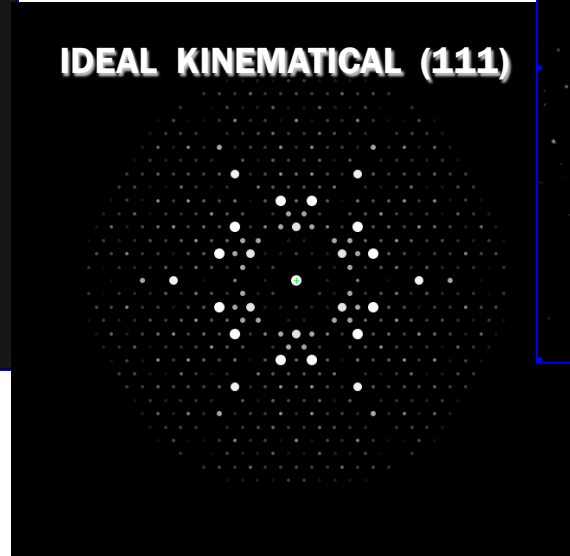
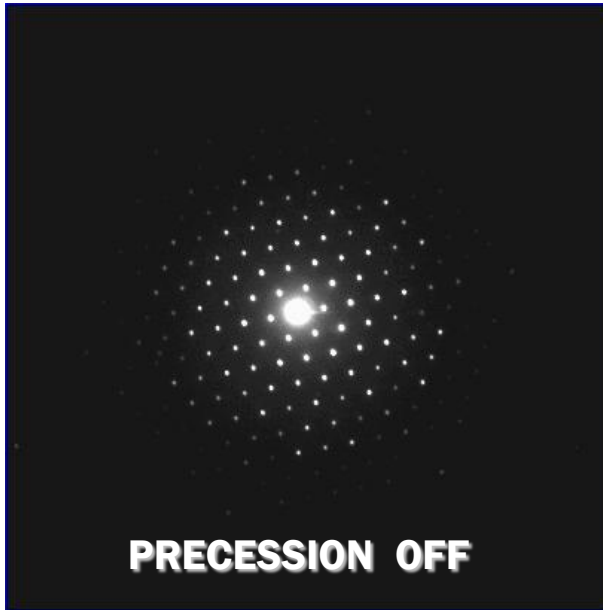
**Jeol 2010F**

**Tecnai 20F**

**Tecnai 30F**



# PRECESSION : QUASI – KINEMATICAL (X-Ray like) INTENSITIES



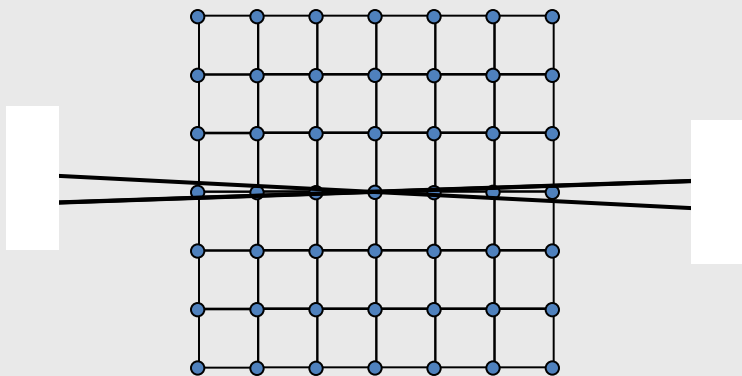
**UVAROVITE cubic mineral (111)**



When applying precession, dynamical conventional SAED patterns ( left ) they become very close to kinematical ( right ) ; compare with simulated kinematical intensities pattern ( center ) .

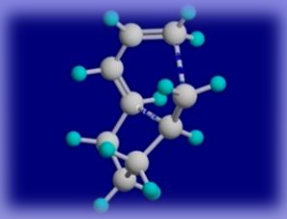
Observe the film of cubic mayenite mineral along 111 ZA , ED pattern how it changes from dynamical to very kinematical at increasing precession angle

# Advantages of precession in single exposure data collection



with beam precession,  
Ewald sphere  
also precess though  
the reciprocal space

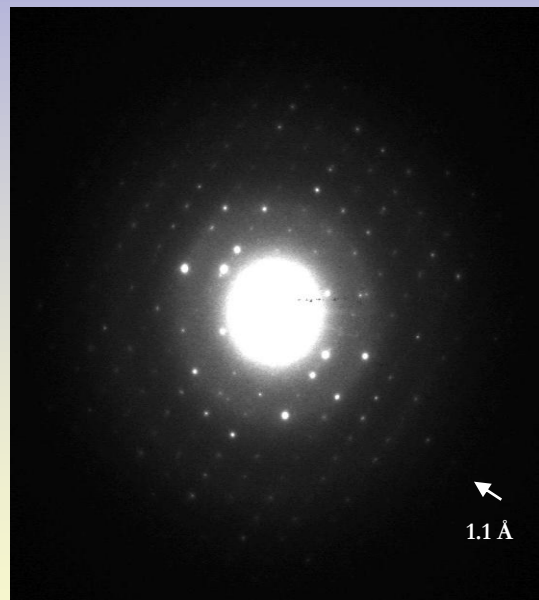
- More fully recorded reflections
- More spots per image
- **Reduced dynamic effect**



# Precession from pharmaceutical nanocrystals

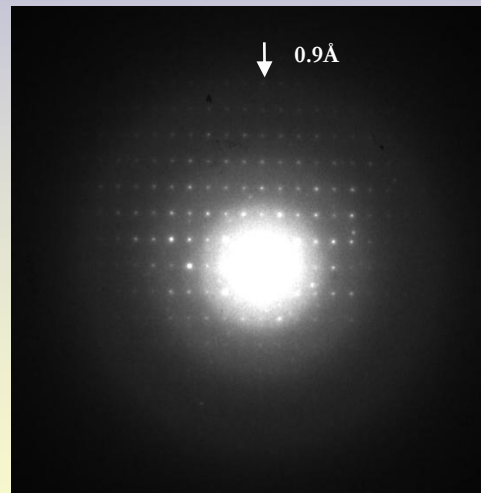
PED patterns in pharmaceutical crystals allow to work with close or with ZA oriented patterns, revealing true crystal symmetry and kinematical intensities good for structure determinations

## amoxycillin

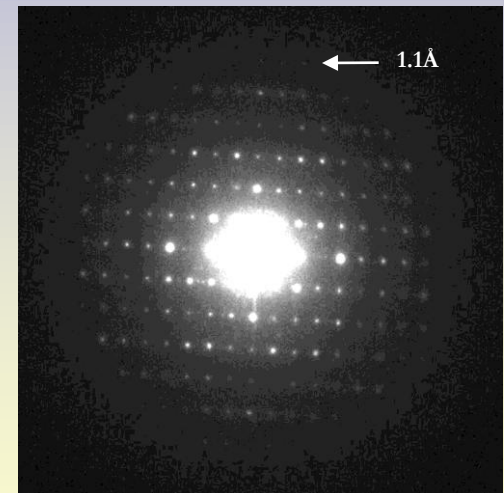


without precession

## penicillin G-potassium



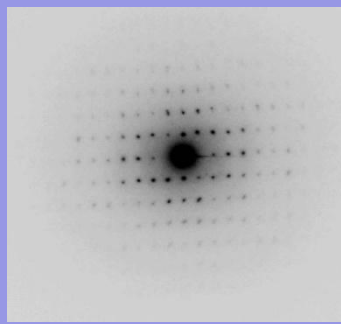
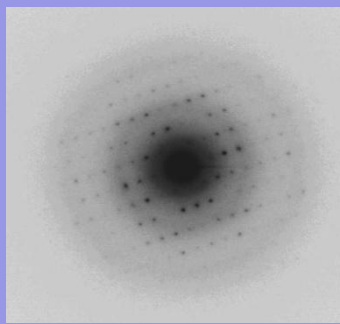
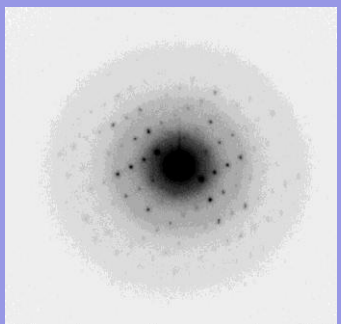
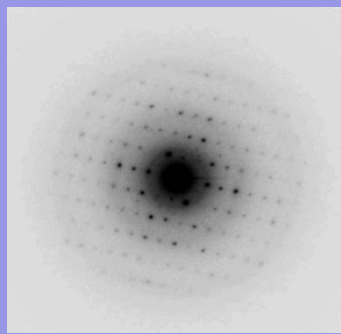
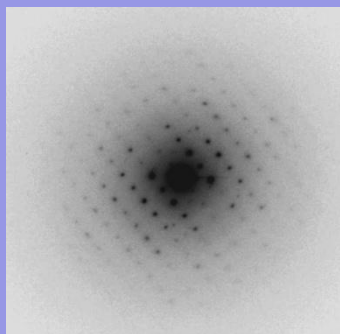
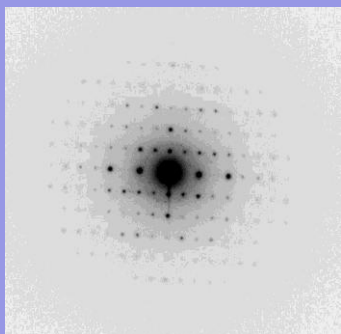
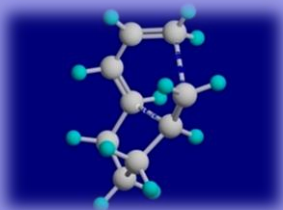
without precession



with precession

# Precession electron diffraction (PED) from penicillin G – potassium

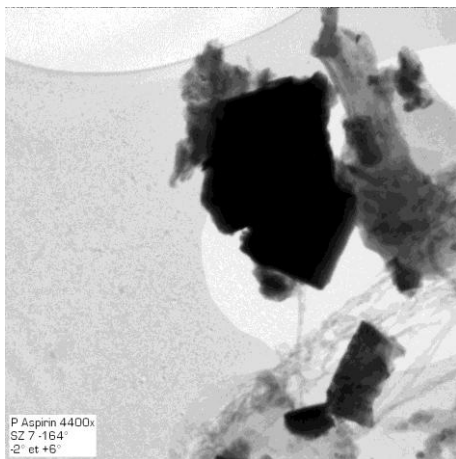
oriented PED patterns show Laue class symmetry



Laue class	Zone axes					
CUBIC CRYSTAL SYSTEM						
m-3m	<111>	<001>	<110>	<uv0>	<uuv>	<uvw>
WP	3m	4mm	2mm	m	m	1
ZOLZ	(6mm)	(4mm)	(2mm)	(2mm)	(2mm)	(2)
m-3	<111>	<001>	<uv0>	<uuv>		
WP	3	2mm	m	1		
ZOLZ	(6)	(2mm)	(2mm)	(2)		
HEXAGONAL CRYSTAL SYSTEM						
6/mmm	<0001>	<11-20>	<1-100>	<uv.0>	<uu.w>	<u-u.w>
WP	6mm	2mm	2mm	m	m	1
ZOLZ	(6mm)	(2mm)	(2mm)	(2mm)	(2mm)	(2)
6/m	<0001>	<uv.0>	<uv.w>			
WP	6	m	1			
ZOLZ	(6)	(2mm)	(2)			
TRIGONAL CRYSTAL SYSTEM						
-3m	<0001>	<11-20>	<u-u.w>	<uv.w>		
WP	3m	2	m	1		
ZOLZ	(6mm)	(2)	(2mm)	(2)		
-3	<0001>	<uv.w>				
WP	3	1				
ZOLZ	6	2				
TETRAGONAL CRYSTAL SYSTEM						
4/mmm	<001>	<100>	<110>	<u0w>	<uv0>	<uvw>
WP	4mm	2mm	2mm	m	m	1
ZOLZ	(4mm)	(2mm)	(2mm)	(2mm)	(2mm)	(2)
4/m	<001>	<uv0>	<uvw>			
WP	4	m	1			
ZOLZ	(4)	(2mm)	(2)			
ORTHORHOMBIC CRYSTAL SYSTEM						
mmm	<100>	<010>	<001>	<0vw>	<u0v>	<uvw>
WP	2mm	2mm	2mm	m	m	1
ZOLZ	(2mm)	(2mm)	(2mm)	(2mm)	(2mm)	(2)
MONOCLINIC CRYSTAL SYSTEM (unique axis b)						
2m	<010>	<u0w>	<uvw>			
WP	2	m	1			
ZOLZ	(2)	(2mm)	(2)			
TRICLINIC CRYSTAL SYSTEM						
-1	<uvw>					
WP	1					
ZOLZ	(2)					

**Easier to find crystal cell parameters from “randomly oriented” PED patterns**  
**Orthorombic a = 6.4 Å b = 9.4 Å c = 30 Å**

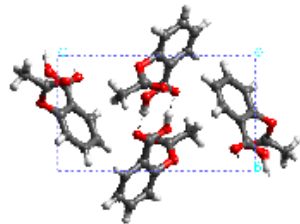
# Obtaining « randomly oriented » PED in PHARMACEUTICALS



## ASPIRIN

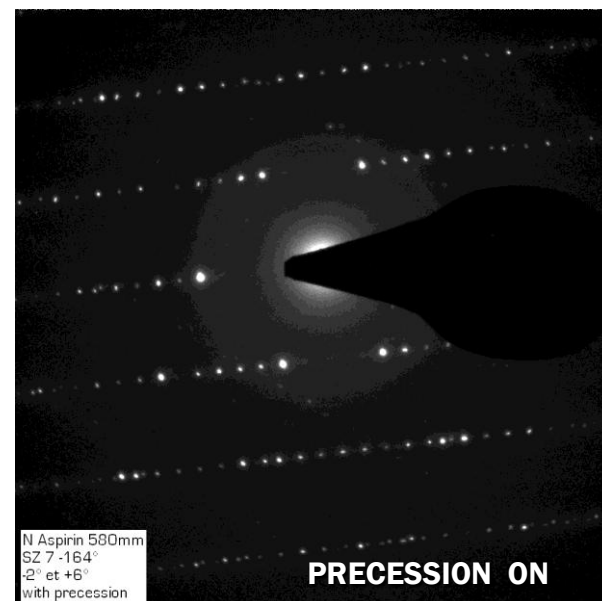
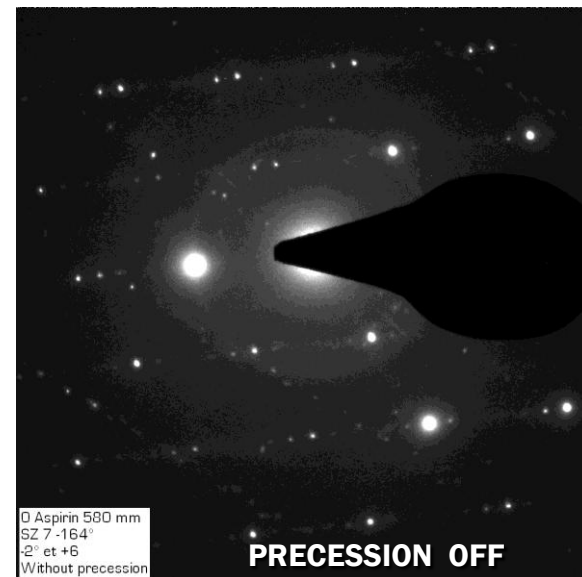
### Potential Polymorphs of Aspirin

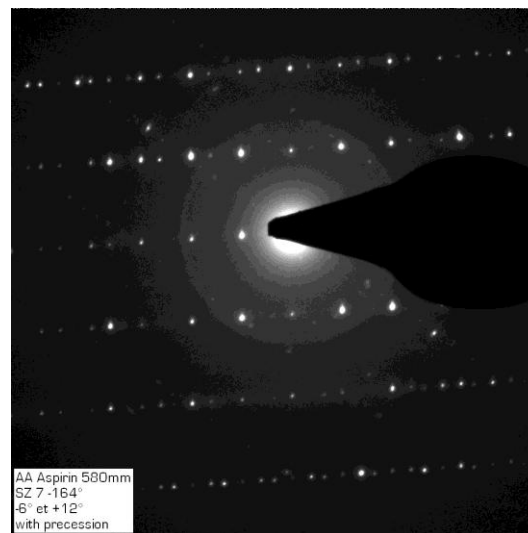
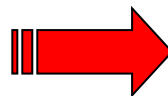
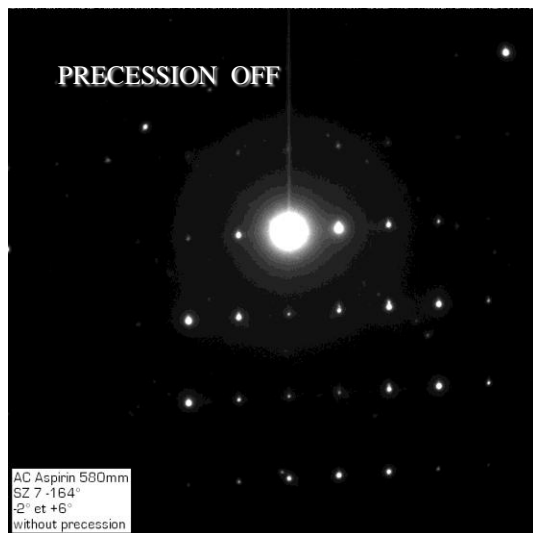
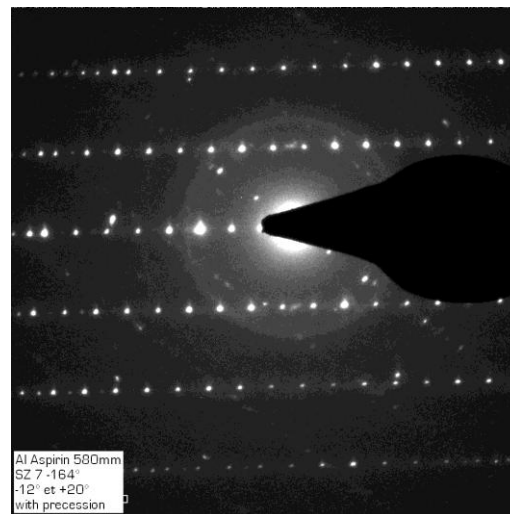
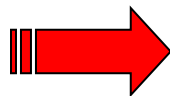
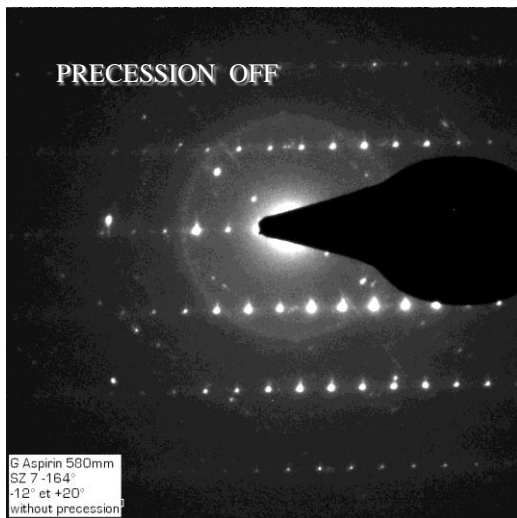
The Polymorph Predictor was used to examine the potential for additional polymorphs of aspirin.



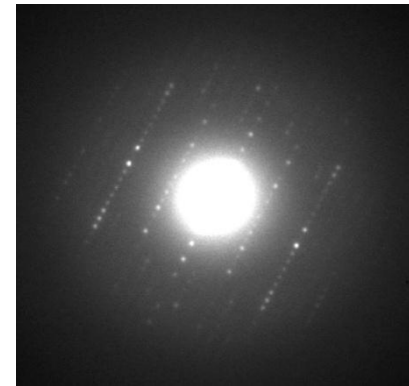
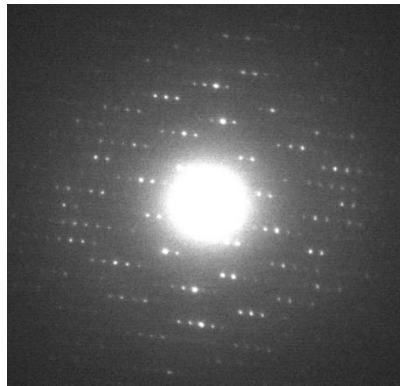
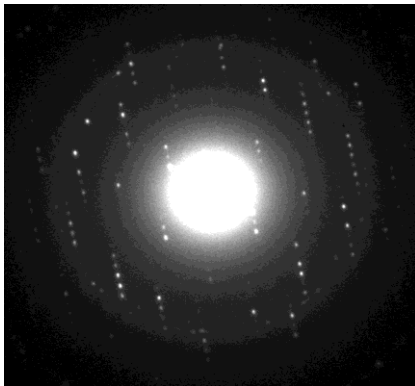
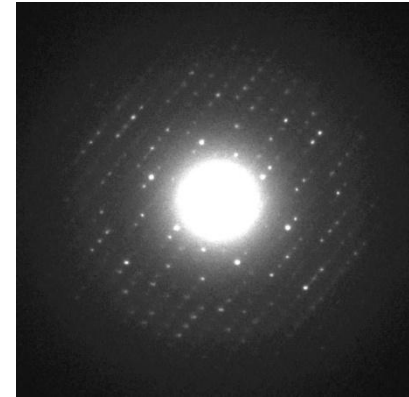
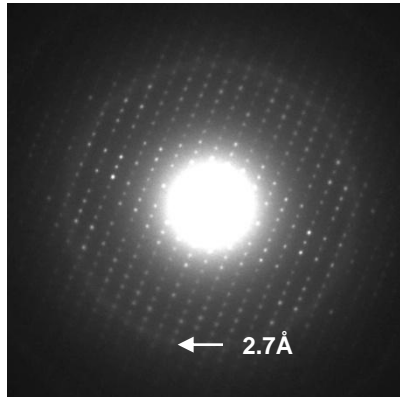
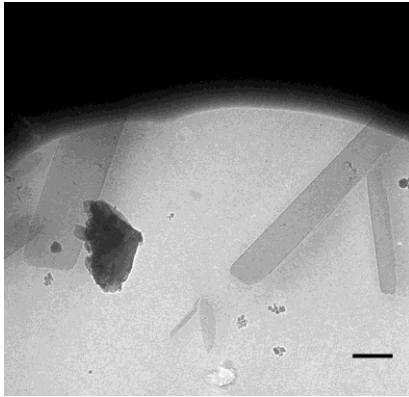
The crystal structure for aspirin. Click on it for a more detailed picture.

Aspirin, also known as acetylsalicylic acid, has numerous pharmaceutical applications. First synthesized in 1897, aspirin is only found experimentally in



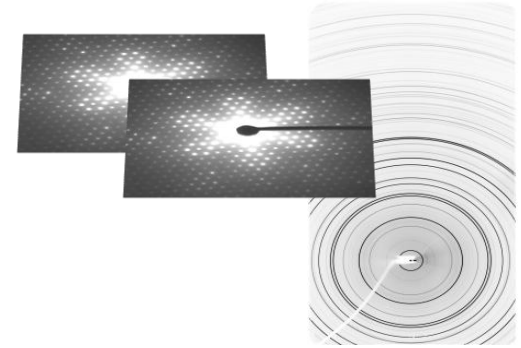
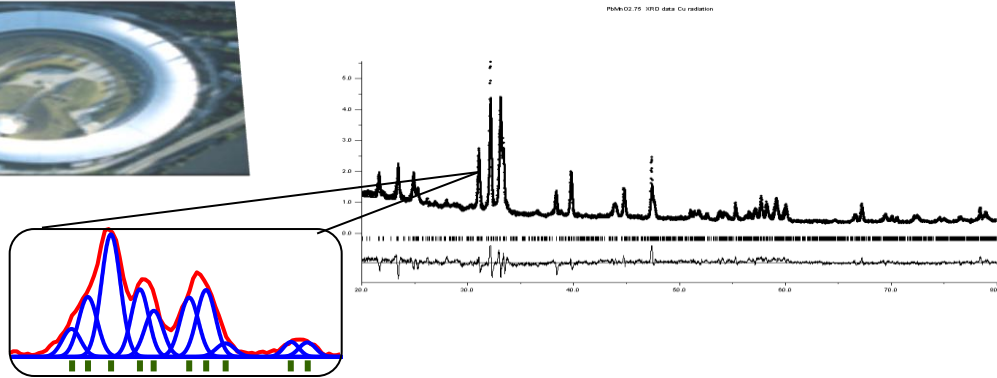


# Data collection : precession electron diffraction from lysozyme nanocrystals $P4_32_12$ $a=b=79.2$ Å , $c=38.0$ Å



Courtesy JP Abrahams, D.Georguieva Univ Leiden

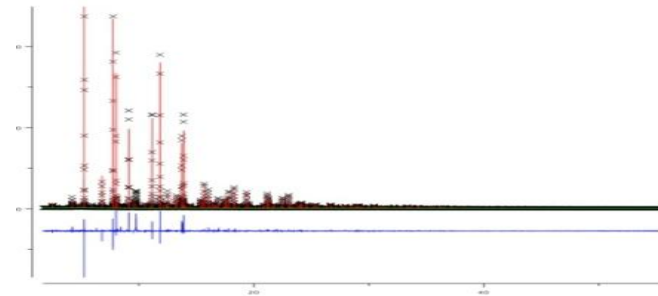
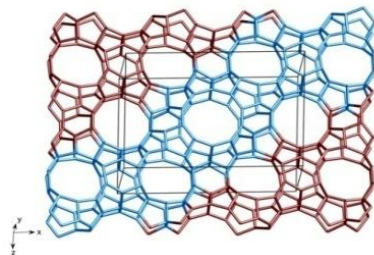
# Combine precession electron diffraction - powder X-Ray diffraction to solve complex structures



Information from PED can be combined with *hkl reflections from X-Ray powder diffraction* to accurately solve and refine *ab-initio* structures using either *charge-flipping* algorithms or direct methods.

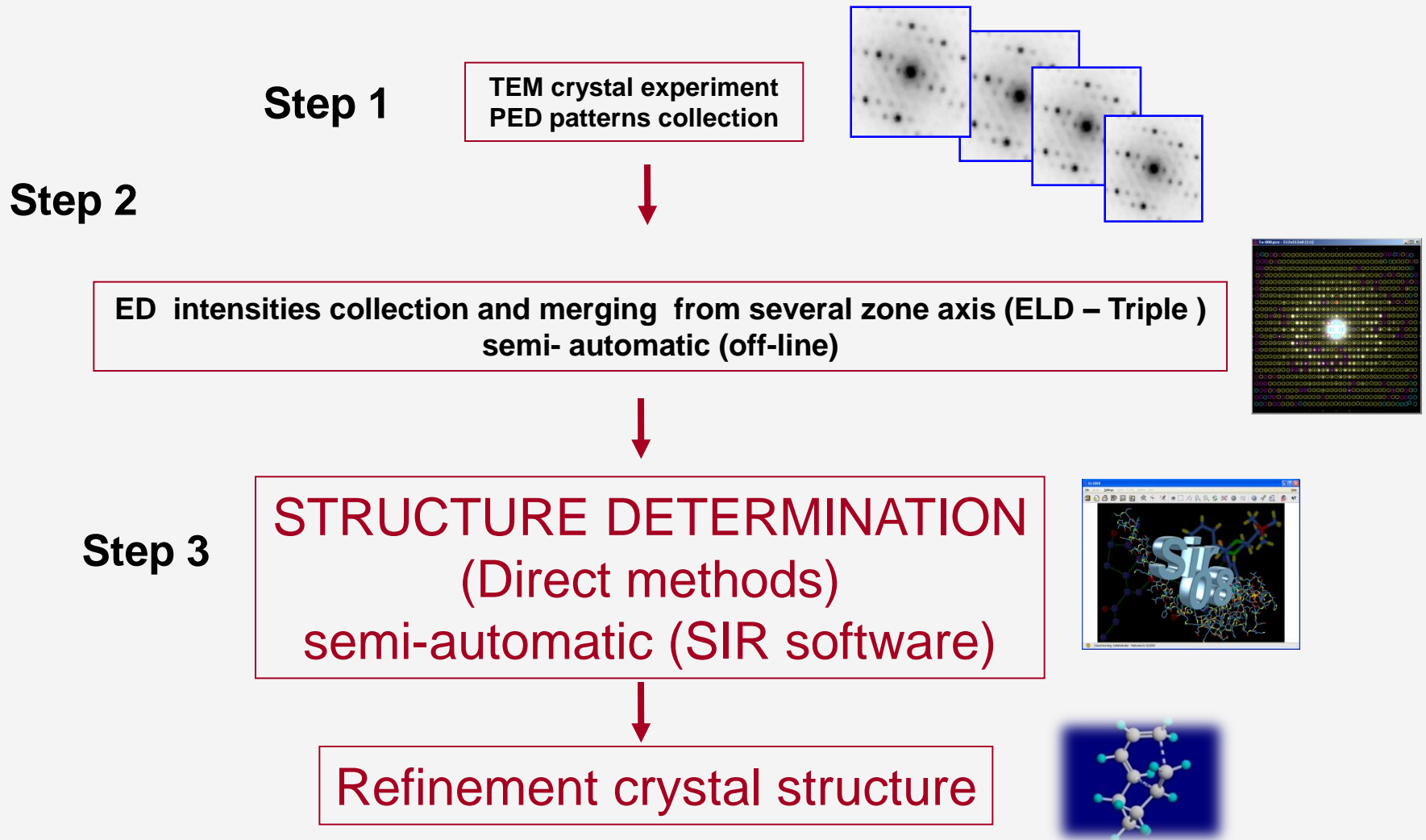
- technique is useful for poorly crystallized / nm size polycrystalline materials
- useful for solving structures of complex organic and inorganic materials
- useful in presence of unknown phases in X-Ray powder pattern

one of the most complex known zeolites TNU-9 ( $\text{Si}_{19}\text{O}_{234}$ ) has been solved by combining Synchrotron X-Ray powder results and data from 5 ZA PED patterns from 300 kV TEM





# Precession electron diffraction : *ab initio* determination of nanostructures



# ***Precession electron diffraction : steps to solve crystal structure***

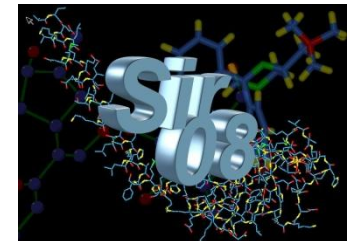
**STEP 1 : collect precession diffraction patterns from oriented ZA; for symmetrical crystals ( cubic, tetragonal ) a few 3-4 patterns may be enough ; collection can be done with films ( less precise method as films are easily saturated in intensity , image plates , CCD or our electron diffraction dedicated electron diffractometer )**

**STEP 2 : extract electron diffraction intensities automatically by software; merging intensities from different ZA by comparing and establish scale factor between common row intensities**

**STEP 3 : after reducing intensities , considering possible space group symmetry , input HKL and intensities at direct methods software using electron diffraction scattering factors ( example SIR2008 , SHELX etc.. ) to solve structure**

**STEP 4 : all atomic positions and atomic type will appear as one of the most probable solutions; heavy atoms are usually all placed in correct positions, while lighter atoms ( eg oxygen appear displaced from ideal positions ).**

R crystallographic residual from PED intensities is usually between 10-25 % as structure solution by electron diffraction is less precise than X-Ray solution where R is 3-5%



# Electron diffraction intensities are measured automatically

FROM

Image plates

Photo film

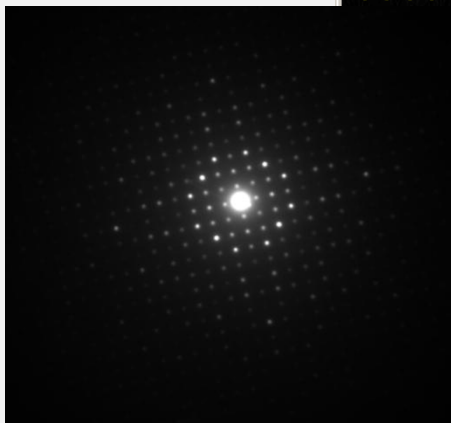
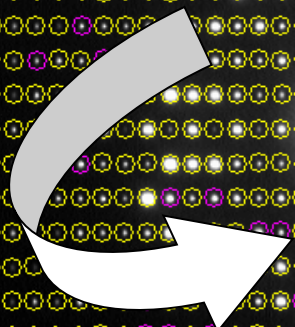
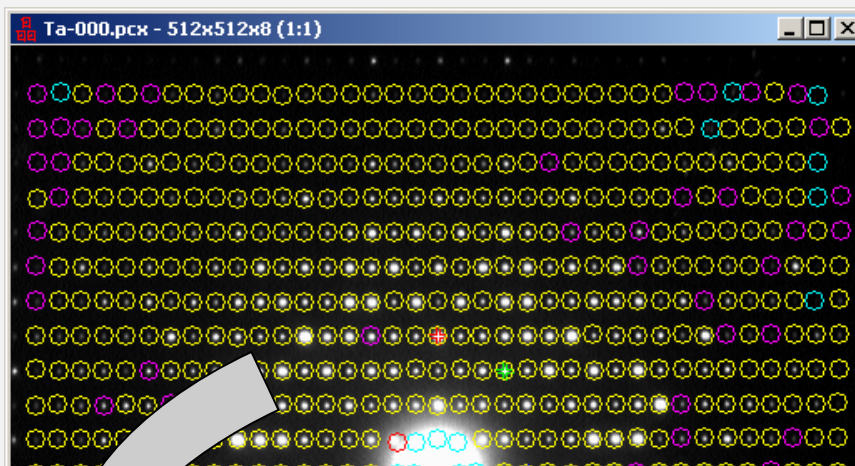
OR

CCD camera

1k x 1k

2k x 2k

4k x 4k



Triple

File EditGroup Tools List Window Help

zr7zse.hke, 825 reflections in P\_1

Format: h k l a d

h	k	l	a	d
0	-2	0	148.6	7.92
0	178.2	7.92	0	0
0	232.2	6.68	0	0
0	173.2	6.68	0	0
0	213.6	6.66	0	0
0	190.6	6.55	0	0
0	538.5	6.20	0	0
0	461.5	6.20	0	0
0	86.8	5.78	0	0
0	62.7	5.78	0	0
0	302.3	5.77	0	0
0	68.9	5.77	0	0
0	83.0	5.28	0	0
0	13.8	5.28	0	0
0	157.9	4.89	0	0
0	173.2	4.89	0	0
0	200.9	4.87	0	0
0	150.7	4.87	0	0
0	107.6	4.86	0	0
0	69.7	4.86	0	0
0	88.1	4.85	0	0
0	117.8	4.85	0	0
0	135.0	4.03	0	0
0	199.6	4.03	0	0
0	159.4	4.01	0	0
0	232.5	4.01	0	0
0	171.7	4.00	0	0
0	138.4	4.00	0	0
0	122.5	4.00	0	0
0	146.8	4.00	0	0
0	170.3	3.96	0	0
0	236.0	3.96	0	0
0	477.9	3.77	0	0
0	283.9	3.77	0	0
0	458.7	3.77	0	0
0	546.3	3.77	0	0
0	184.6	3.67	0	0
0	308.9	3.67	0	0
0	261.9	3.66	0	0
0	359.0	3.66	0	0
0	231.5	3.34	0	0

Space Group Explorer V1.0

Search: International Tables Vol.Δ (1983) International Tables Vol.I (1965) Hall

Info: Space Group: Pnmm, Hall symbol: P 2<sub>1</sub>2<sub>1</sub>, Point group: mmm, Lattice group: mmm, Crystal system: Diphonohobic, Lattice symbol: P, primitive, Extra info: None

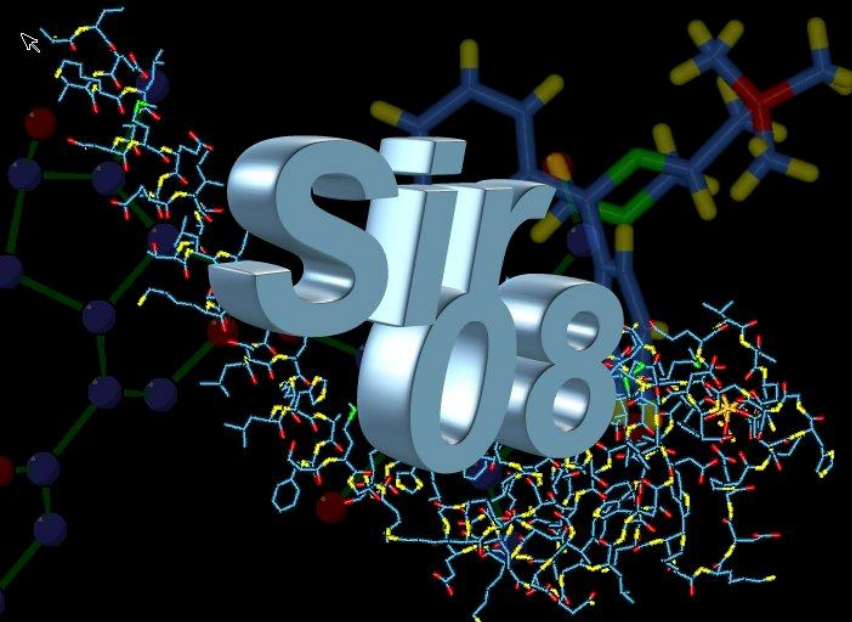
Semivariant vectors: (h,k,l) mod (2,2,2)

Order: 8, Unique axis: Order P: 8

Reflections with phase restriction: 05 hkl: 0, 04 hkl: <80k>h, 03 hkl: <80k>h, 02 hkl: <0>, 01 hkl: <80k>, 08 hkl: <80k>, 07 hkl: h+k=2n, 06 hkl: No conditions, 05 hkl: No conditions, 04 hkl: h=2n, 03 hkl: h=2n, 02 hkl: No conditions, 01 hkl: No conditions

Systematically Enhanced

Reflections with phase restriction: 05 hkl: e=1, 04 hkl: e=2, 03 hkl: e=2, 02 hkl: e=2, 01 hkl: e=4, 08 hkl: e=4, 07 hkl: e=4, 06 hkl: e=4, 05 hkl: e=4, 04 hkl: e=4, 03 hkl: e=4, 02 hkl: e=4, 01 hkl: e=4



Use Direct methods, charge flipping  
(like X-Ray) to solve crystal structures

## DIRECT METHODS (electron scattering )

SIR 96, SIR 2007, SIR 2008, FOX ,.....

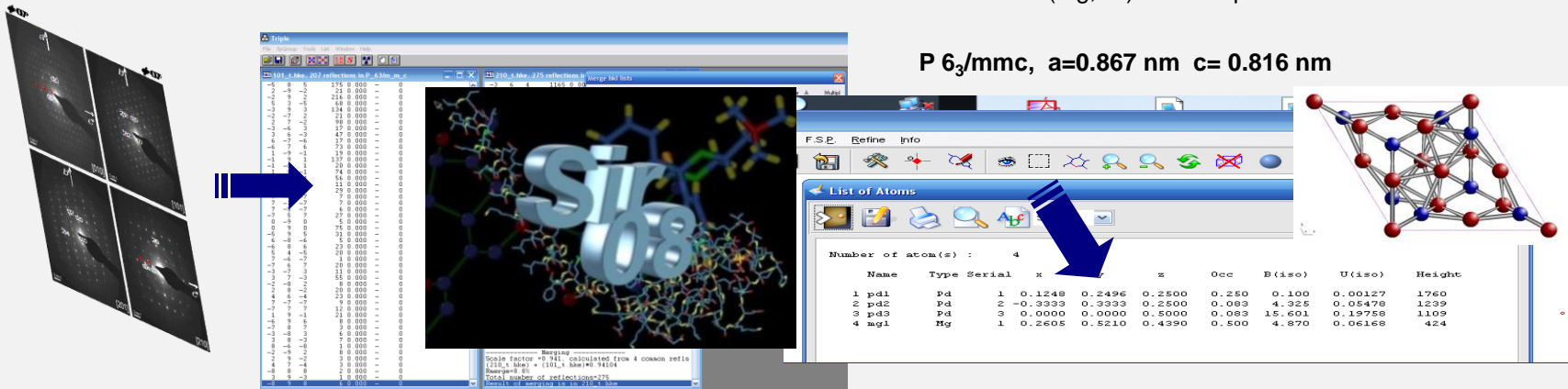
**Structure solution with direct methods: SIR2008**

[http://www.ic.cnr.it/registration\\_form.php](http://www.ic.cnr.it/registration_form.php)

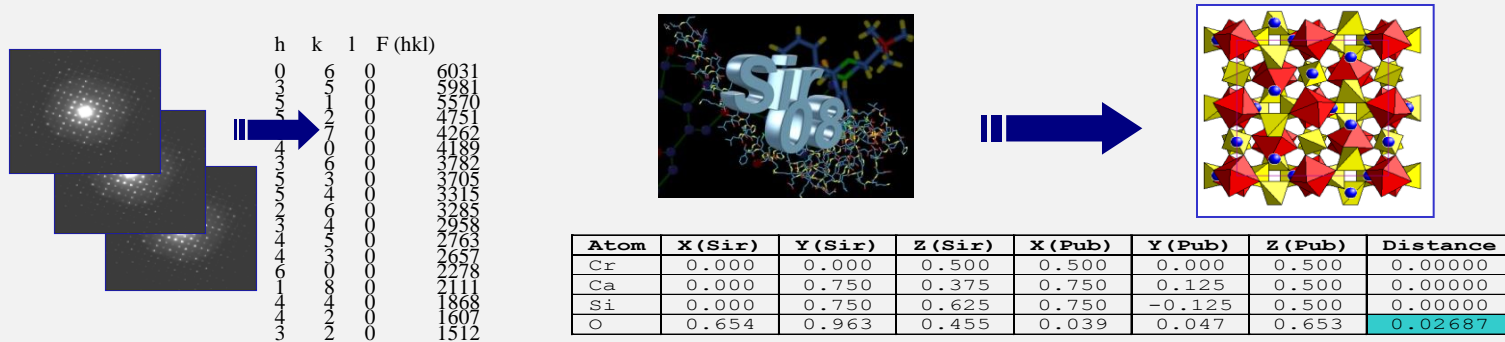
FullProf suite : <http://www.ill.eu/sites/fullprof/>

# Use precession diffraction intensities to solve crystal structures

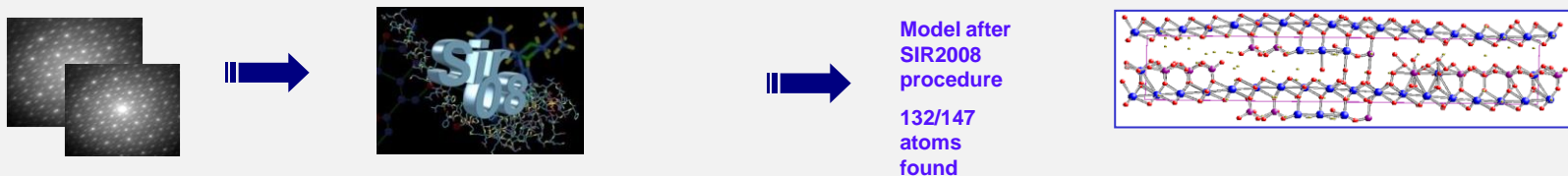
In this example PED intensities from 5 zone axis (ZA) from **Mg<sub>5</sub>Pd<sub>2</sub> nanocrystal** have been collected with a 100 kV TEM. Use of **SIR2008** direct methods software allowed direct calculation of all the (Mg,Pd) atomic positions.



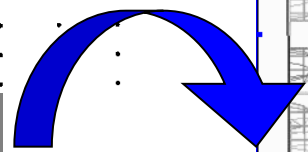
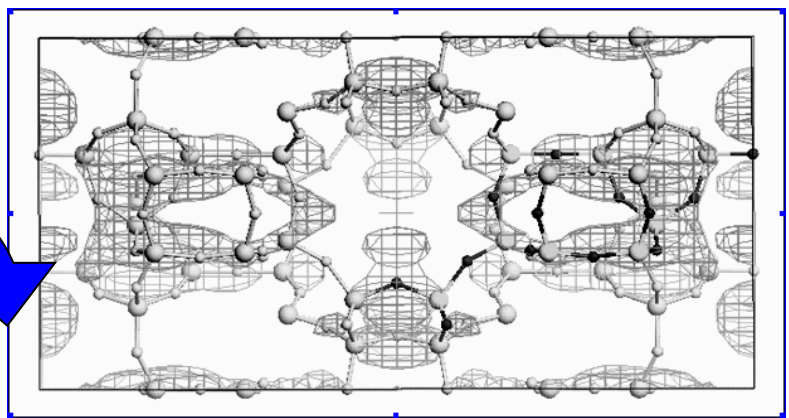
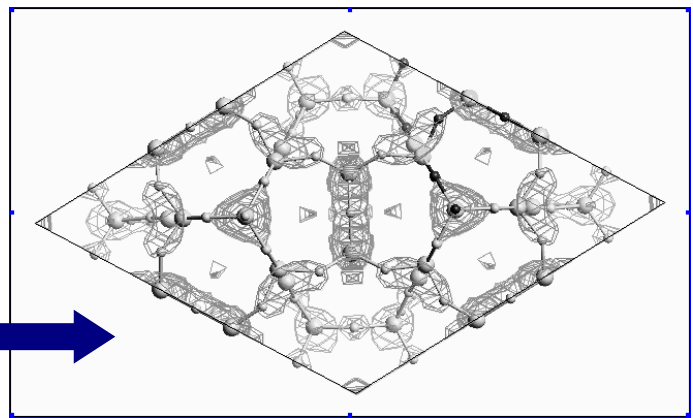
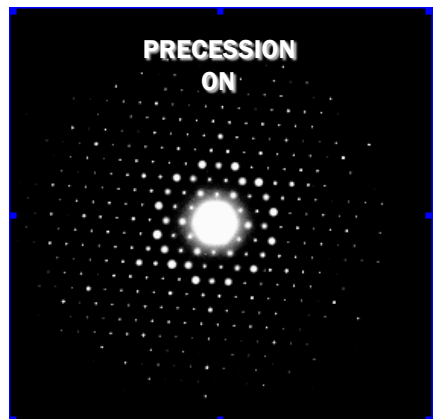
Collection of **3 ZA** (**[0 0 1]**, **[1 0 1]**, **[1 0 2]**) **PED intensities** with a 200KV TEM from **uvavovite mineral Ca<sub>3</sub>Cr<sub>2</sub>(SiO<sub>4</sub>)<sub>3</sub> cubic Ia-3d a=1.2nm** and the use of SIR2008 revealed precise atomic structure (see table) **calculated atomic positions are very close to X-Ray 3D refined atomic model**



Using set of simulated PED intensities from 2 ZA of complex commensurate structure of **antigorite mineral ~Mg<sub>3</sub>Si<sub>2</sub>O<sub>5</sub>(OH)<sub>4</sub>** and solving with direct methods (SIR2008), **most of the atomic positions are revealed**



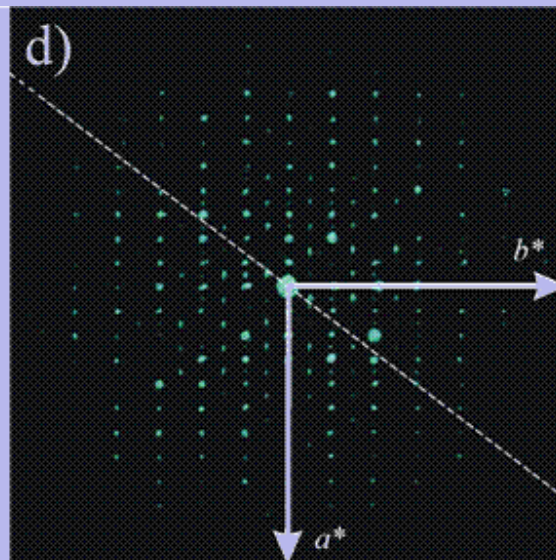
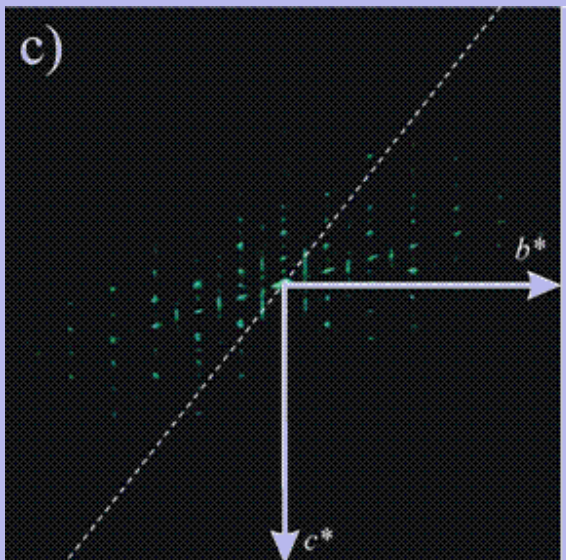
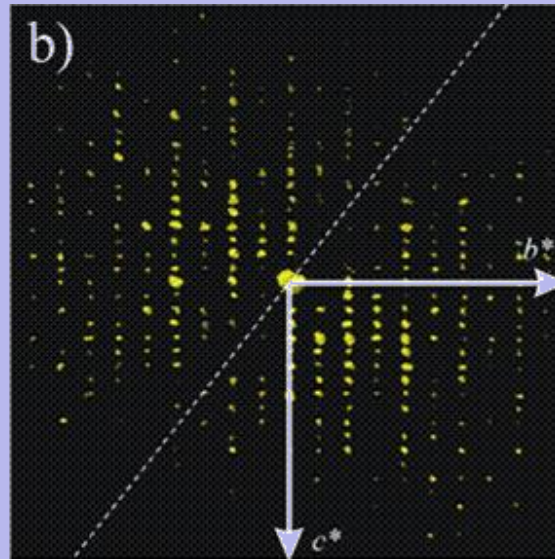
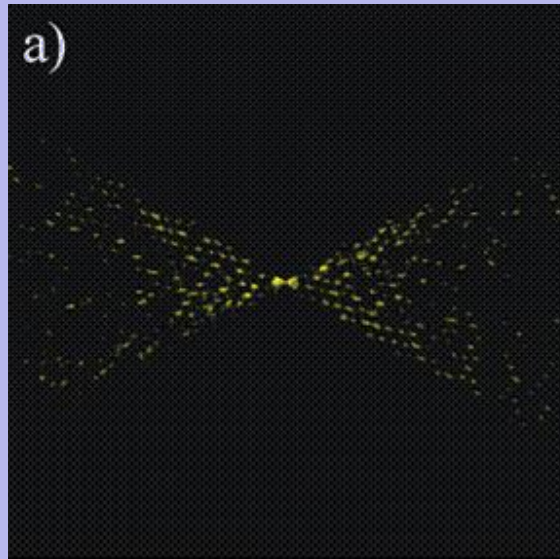
# Ab initio determination of MCM-22 (ITQ-1) zeolite framework



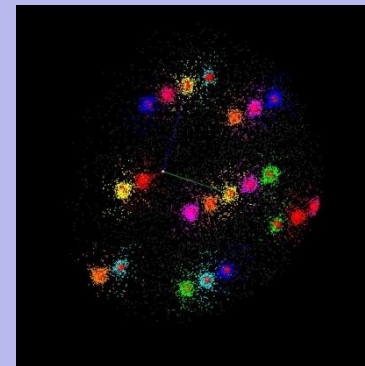
3D frameworks can be revealed by collecting and combining quasi-kinematical precession electron diffraction intensities from different zone axis to one 3D electron diffraction data set ( image courtesy Douglas Dorset USA)

06	15	25	35
04	14	24	34
03	13	23	33
02	12	22	32
01	11	21	31
	10	20	30

# 3D precession electron diffraction tomography (ADT) 3D visualization - Detection of disorder



Cell parameter determination  
on single  
nanoparticle



Conference by Dr. Gorelik

Courtesy : Prof. U Kolb UMainz

# Characterization

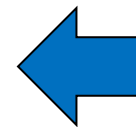
for new unknown compound



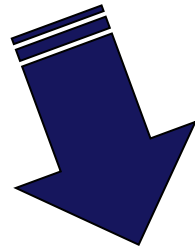
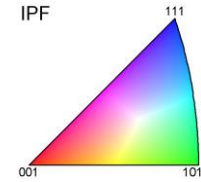
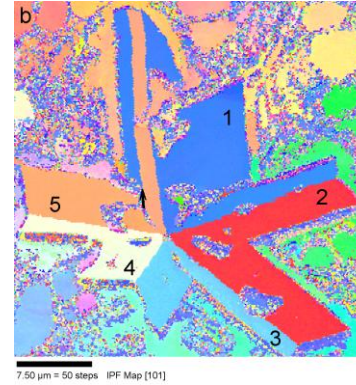
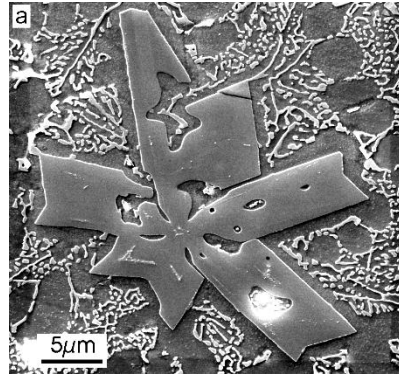
**Find crystal cell parameters**

**Find atomic crystal structure**

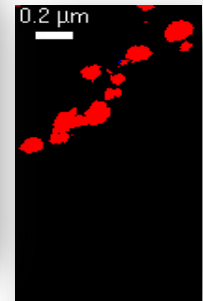
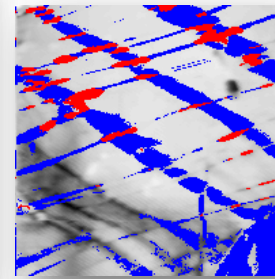
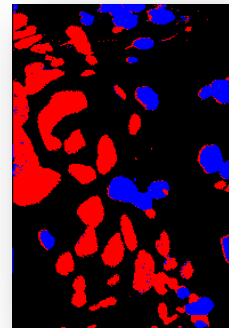
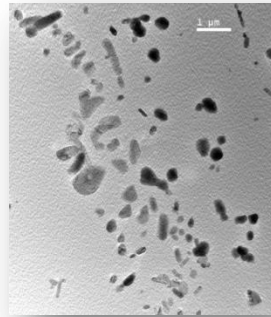
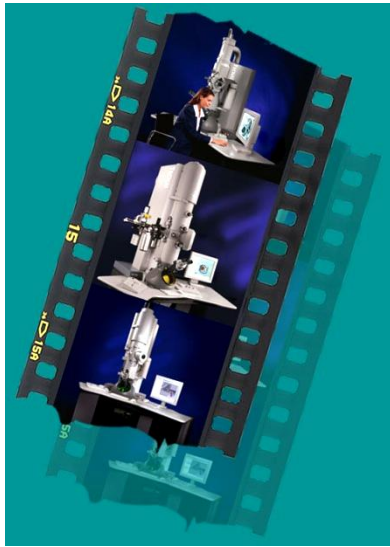
**Texture – (multi) phase analysis**



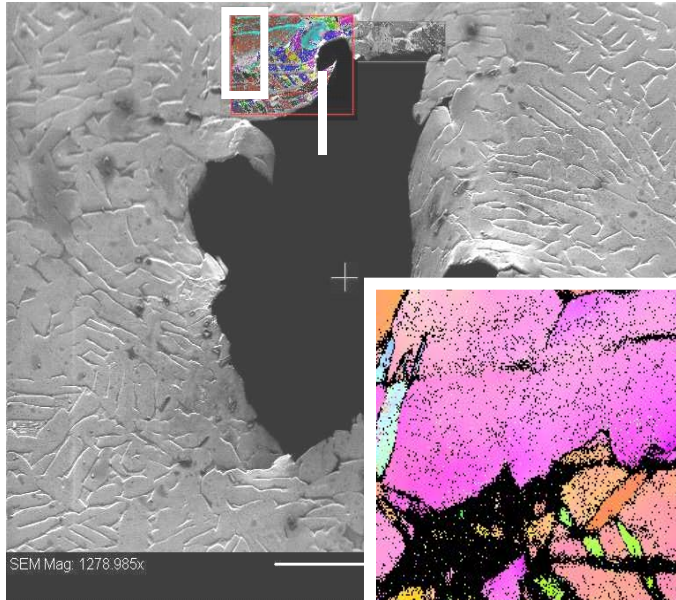




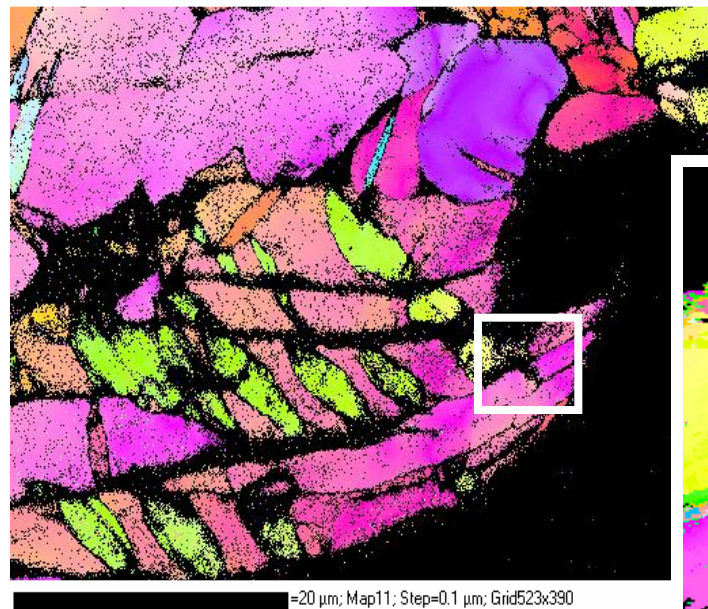
## NEW precession application “EBSD” – TEM



# Comparison SEM-(EBSD) vs TEM spatial resolution

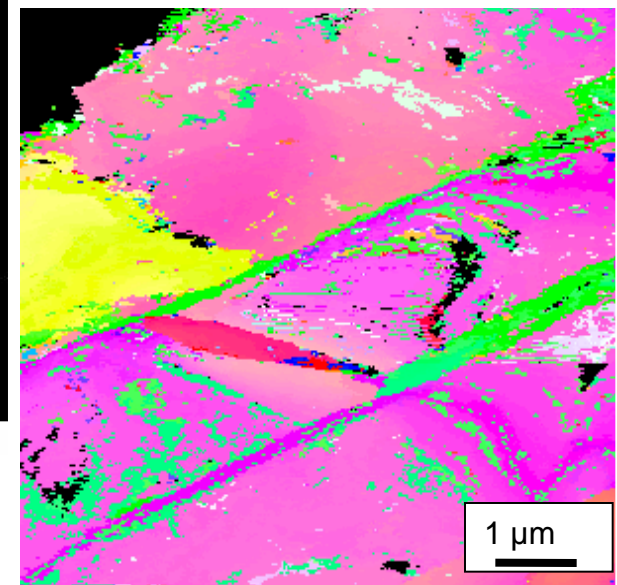


**SEM orientation map  
deformed Ta6V alloy**



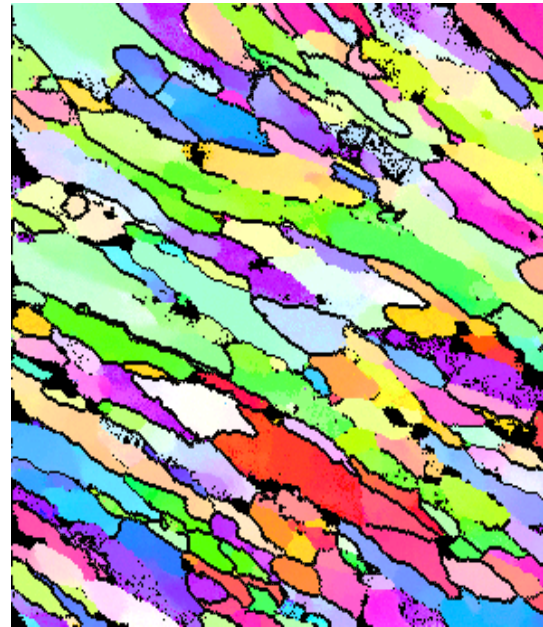
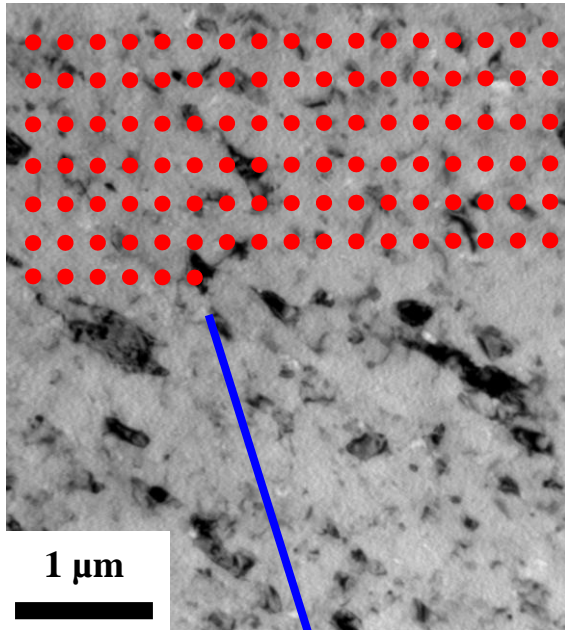
**EBSD-SEM map ( 50 nm resolution)**

**TEM orientation map  
(1-10 nm stepsize)**

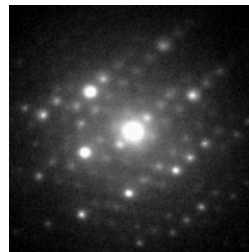


**Electron Backscattering Diffraction (EBSD ) orientation maps in SEM have usually poor resolution in comparison with TEM maps showing detailed nanostructure**

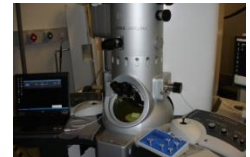
# ASTAR : diffraction pattern adquisition



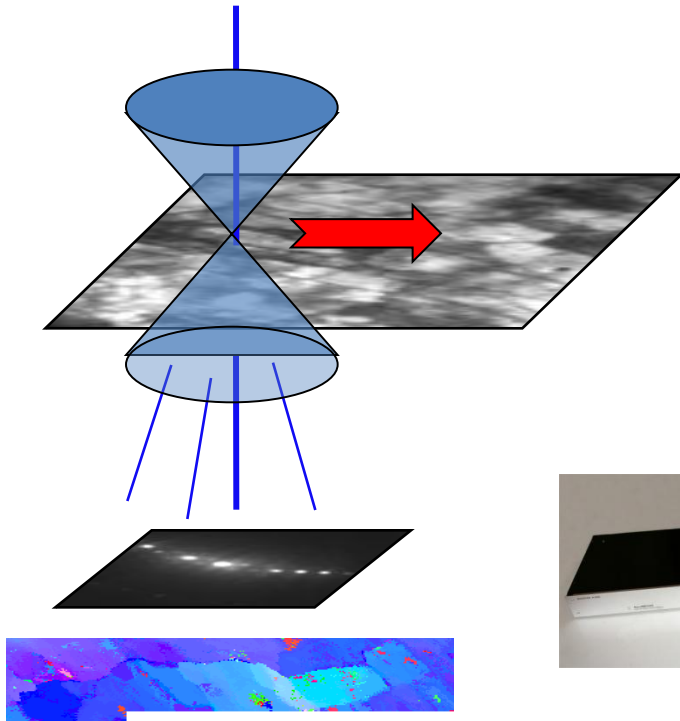
Example :Severely  
deformed  
7075 Aluminium Alloy



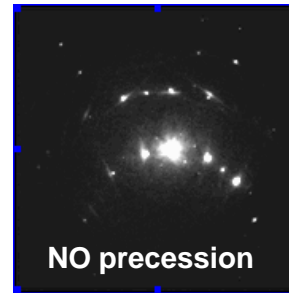
Any TEM -FEG/LaB6  
may work with ASTAR



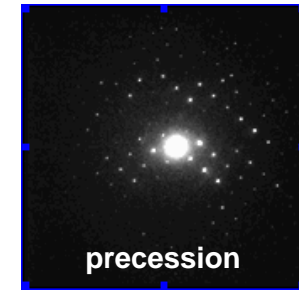
# ASTAR ( EBSD-TEM like procedure )



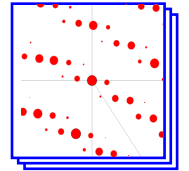
Orientation map



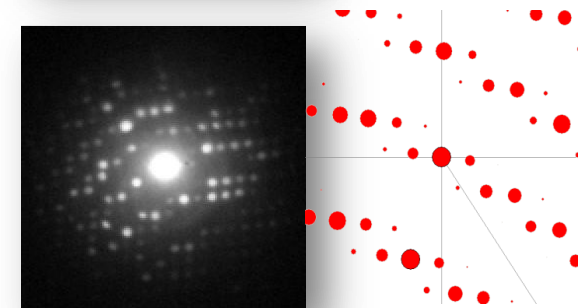
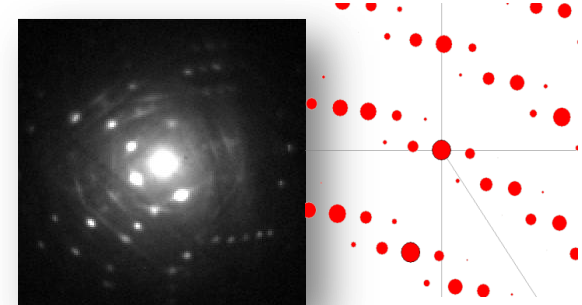
NO precession



precession

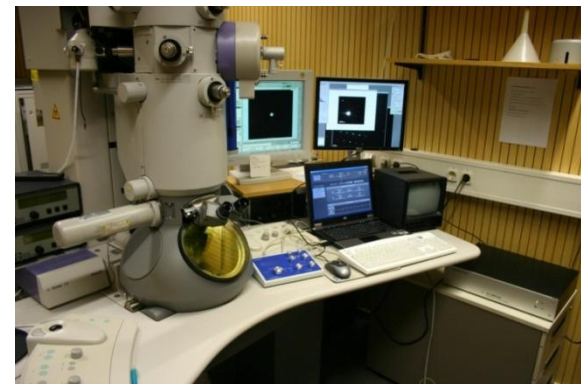
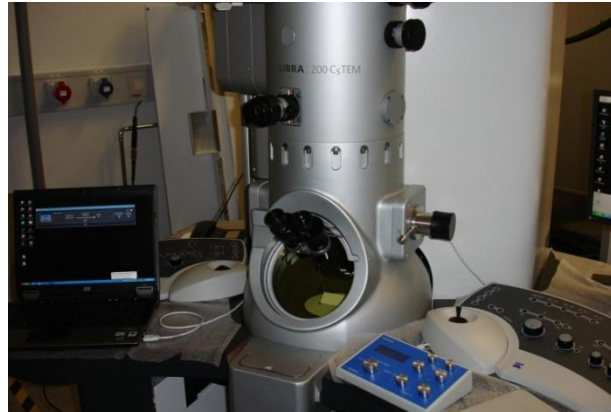
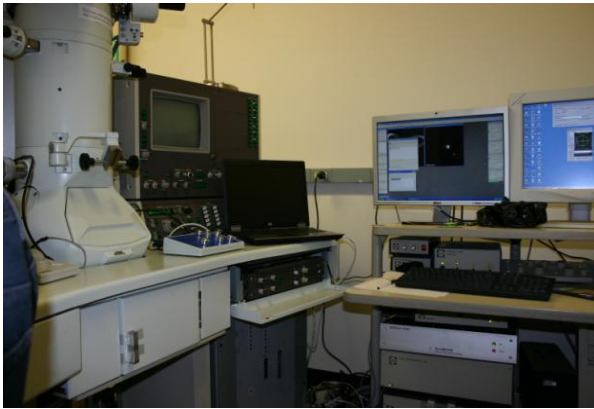


Using precession diffraction the number of ED spots observed increases ( almost double ) ; correlation index map becomes much more reliable when compared with templates



In this example (right) a metal particle gives wrong correlation index without precession due to presence of Kikuchi lines; after applying precession (right lower image), index gets correct value as ED quality improves and Kikuchi lines disappear

# ASTAR phase –orientation mapping for advanced TEM



**Zeiss LIBRA 200F Cs corr**

**TECNAI 20F TECNAI 30F**

**JEOL 2200 FS**

**JEOL 2010F**

# DiffGen : Template generator

The screenshot displays two windows from the DiffGen software. The 'Diffraction pattern' window on the left shows a 2D diffraction pattern with red spots on a grid. The 'Cell structure editor' window on the right shows the crystallographic data for Aluminium oxide (2.667/4) - gamma'.

**Diffraction pattern window:**

- Phi1: 319.79, PhiM: 57.89, Phi2: 62.92
- Orientation: 4 Miller's indices, Approximated hkl: 6 3 4 Err. 1.37°
- Display control: Spot size 5, Count 5137, Zoom level 100, Rotat. step 1
- Symmetry: Laue Class 4/mmm, Current symmetry 0
- Diffraction settings: Wave length (Å) 0.02, Max angle (°) 3, Excitation error 1
- Intensity control: Intensity scale 9, Minimum intensity 0.1

**Cell structure editor window:**

- Structure name: '99836-ICSD 'Aluminium oxide (2.667/4) - gamma''
- Unit cell parameters: a=5.652, b=5.652, c=7.871, alpha=90.00, beta=90.00, gamma=90.00
- Space group: 141 141/a m d, Laue class: 4/mmm
- Atoms count: Base 4, Total 36
- Table of atoms:

	Elem.	Occ	F	x	y	z	Int
1	Al1	0.78	13	0	0.75	0.125	0.780
2	O1	1	8	0	0.0076	0.2516	0.620
3	Al3	0.58	13		0	0.5	0.580
4	Al2	0.36	13		0	0	0.360
5	O1	1			0.75	0.5016	0.620
6							0.20
7							0.20
8							0.20
9							0.20
10							0.20
11	O1	1	8	0.2576	0.75	0.9984	0.620
12	O1	1	8	0.5	0.0076	0.2484	0.620
13	O1	1	8	0.2424	0.25	0.4984	0.620
14	O1	1	8	0.7424	0.75	0.9984	0.620
15	O1	1	8	0.7576	0.25	0.4984	0.620
16	O1	1	8	0.5	0.5076	0.7516	0.620

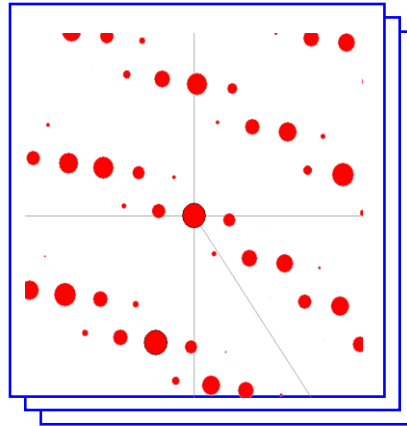
A blue arrow points from the 'Structure name' field to the formula  $F_{hkl} = \sum_i f_i e^{2\pi i (hx_i + ky_i + lz_i)}$ .

Features: Any crystallographic structure  
 Laue class adapted to the space group  
 Structure generator (space group, structure factor equ.)

**input from ICDD database**

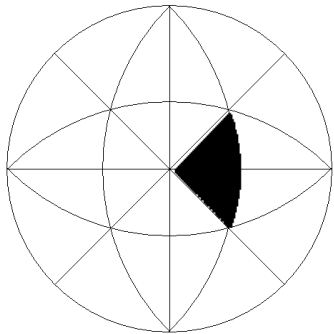
# ASTAR : crystallographic orientation identification

Pre-calculated templates

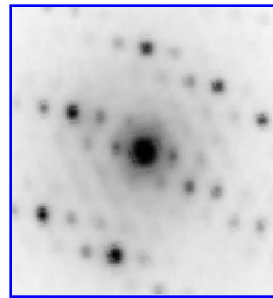


$$Q(i) \sim \sum_{j=1}^m P(x_j, y_j) T_i(x_j, y_j)$$

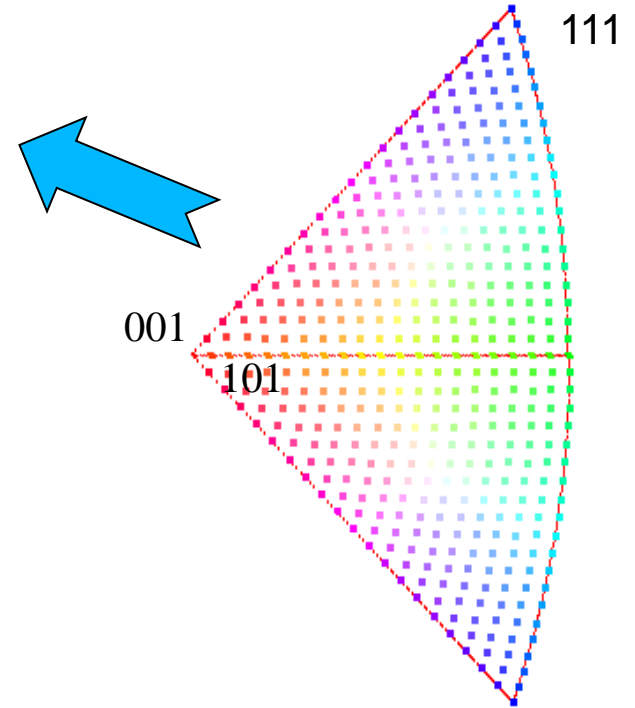
Correlation index



Acquired pattern



Template generation of all possible simulated orientations (every 1°) within stereographic triangle for given crystal lattice(s) and symmetry



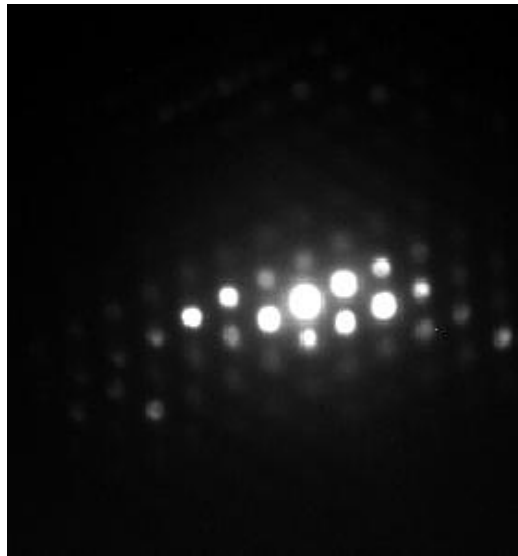
Stereographic projection

1-11

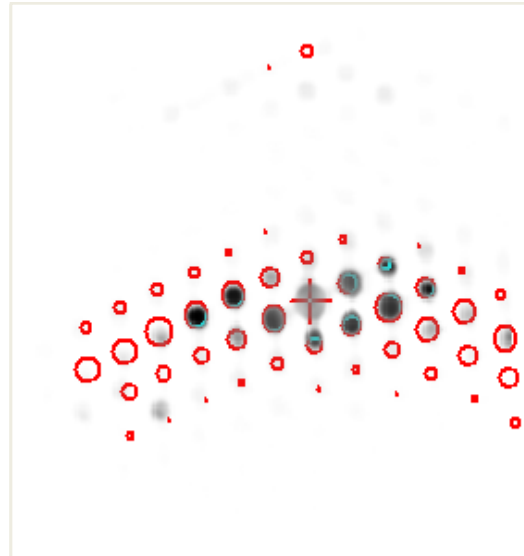
Degree of matching between experimental patterns and simulated templates is given by a correlation index ; highest value corresponds to the adequate orientation/phase

(example ,cubic)~ **2000 simulated patterns**

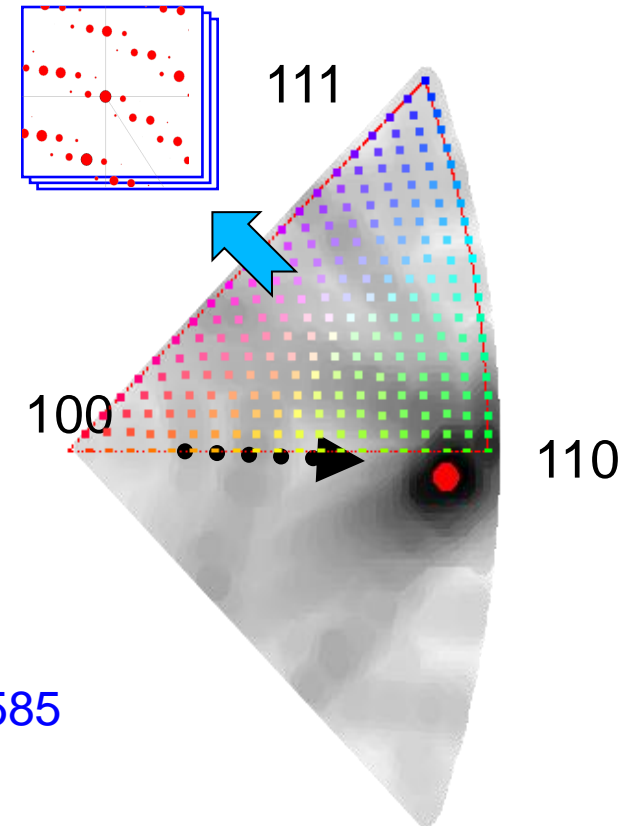
# ASTAR identification example : nanocrystalline Cu



Diffraction pattern  
( nanocrystalline cubic copper)



correlation index = 585



**Correlation index map**

For a given ED pattern, the correlation index map is calculated for all possible template orientations and plotted on a map that represents a portion of the stereographic projection (reduced to a double standard triangle). That resulting map reveals the most probable orientation for every experimental spot ED pattern ( in this case ED pattern is found to be close to 110 ZA orientation )



# ASTAR : ultra-fast TEM orientation map

Sample : severely deformed copper

250 x 200 pixel data acquisition

**5-10 min**

Typical software data analysis  
time ( for cubic )

**5-15 min**

( hexagonal , tetragonal )

x 3- 4 more time

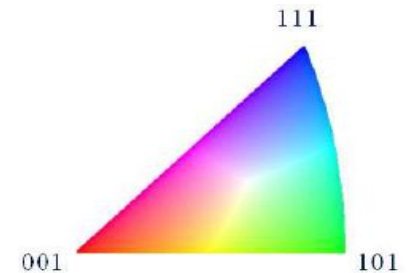
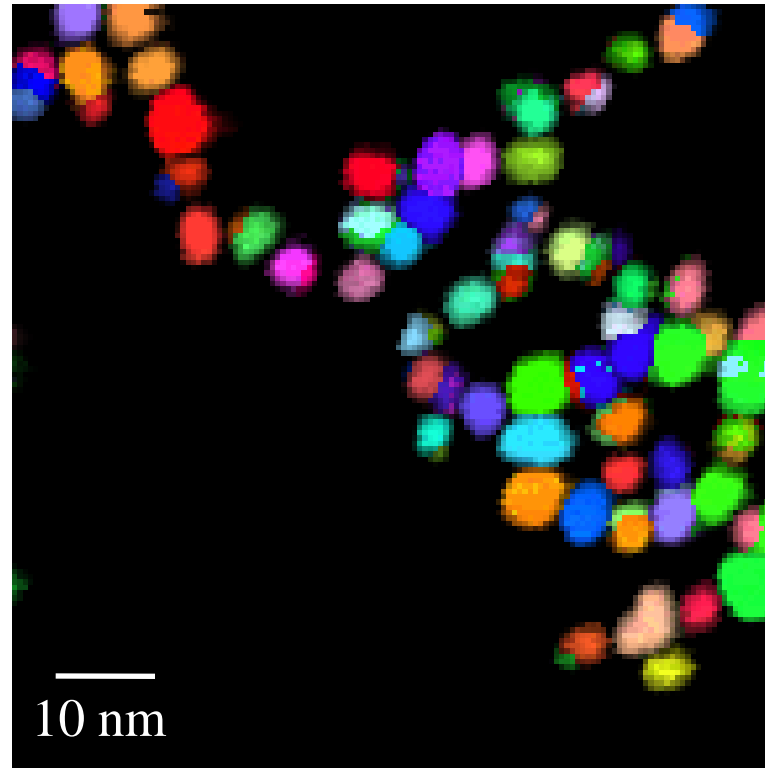


Orientation map

**Map resolution equals beam size resolution**

NBD step 20 nm (LaB6)

# Power of the ASTAR Technique : nanoparticles

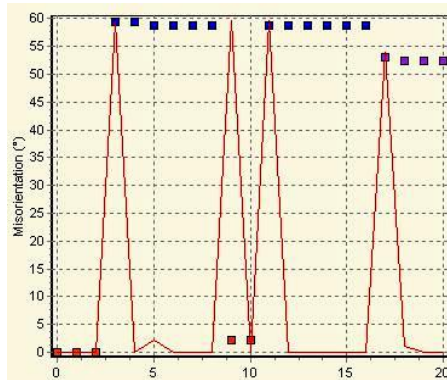
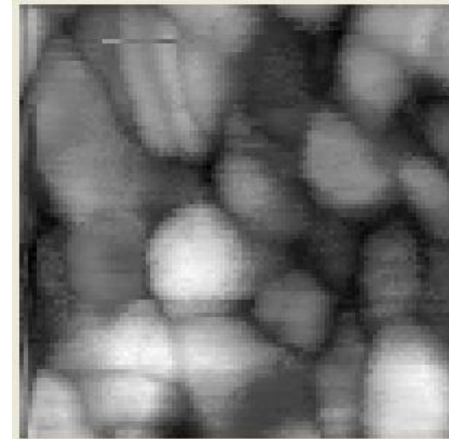
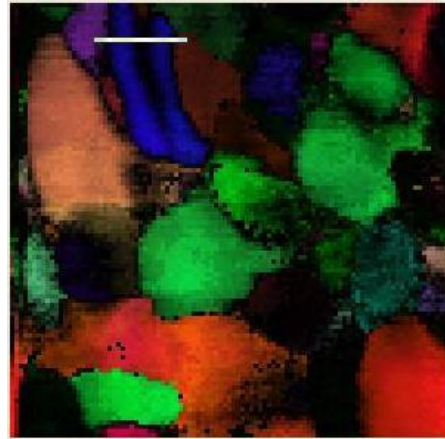
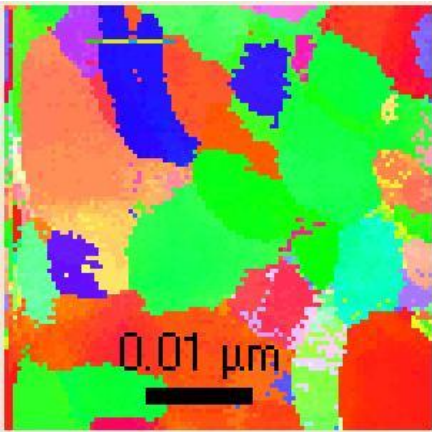


**ASTAR** : Orientation analysis from Pt ~100 particles ~ 6 nm in size

**1 nm resolution for orientation map**

Data courtesy Prof. P.Ferreira, J.Ganesh Univ Texas at Austin USA JEOL 2010 FEG

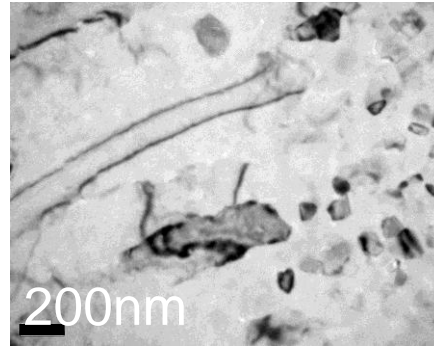
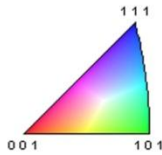
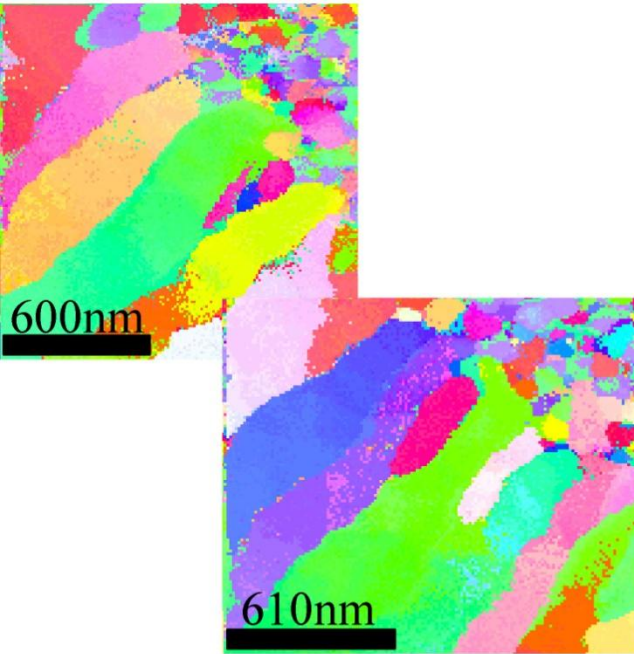




## Pd nanoparticles 0.4 nm twin structure details !

Jeol 2200 FS Humboldt Univ Berlin courtesy Dr. H.Kirmse, I. Heusler

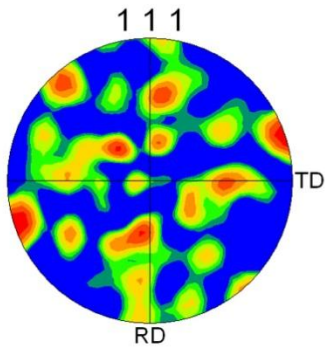
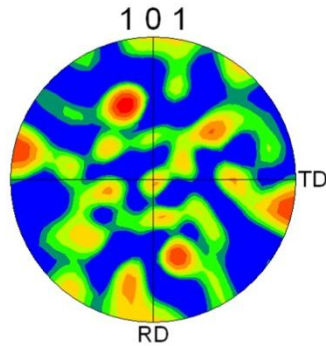
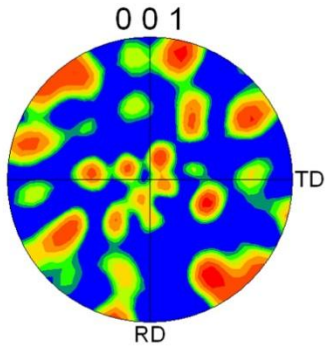
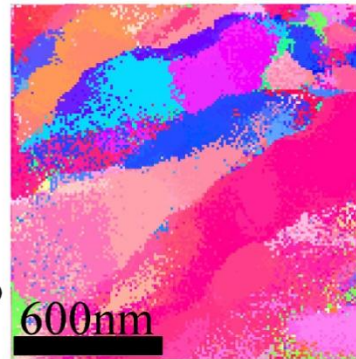
This 8 nm bar is most clearly visible in (b) and divided into 20 sections of 0.4 nm each (that represent the utilized scanning-precession increments of the primary electron beam) in (d). Note that there are jumps in the local mis-orientation profile along the 8 nm bar of almost  $60^\circ$  between sections 3 and 4, 8 and 9, as well as 10 and 11.



**COMPARISON –COMBINATION  
texture analysis  
EBSD-SEM --- EBSD-TEM (ASTAR)**

→ texture analysis of  
several areas (combined)

→ 20 nm to 40 nm step  
size in LaB6-TEM scans  
sufficient to capture  
morphology of  
microstructure with high  
fidelity (ASTAR)

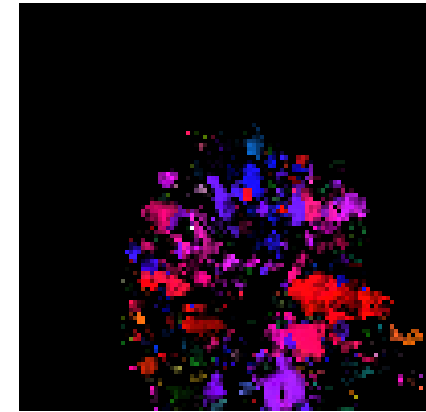
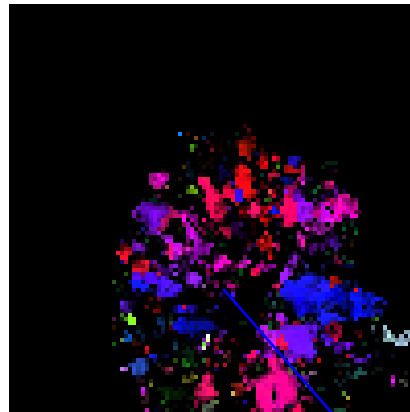
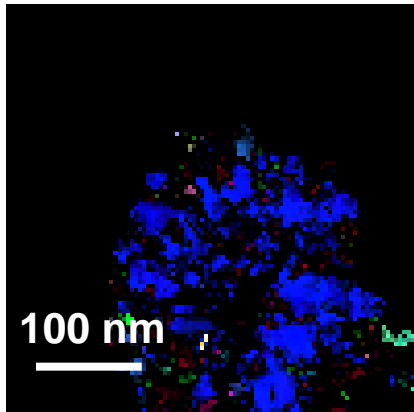


**EBSD-TEM like (ASTAR)**



→ **ASTAR orientation data  
can be read/analyzed by  
TSL –HKL software**

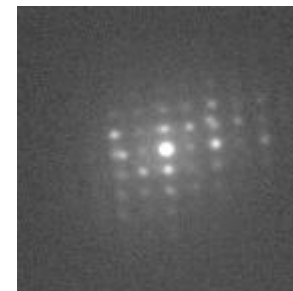
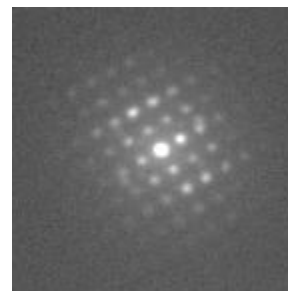
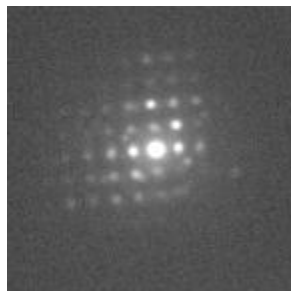
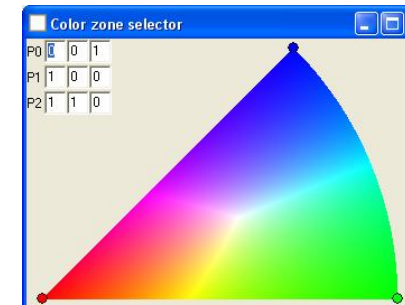
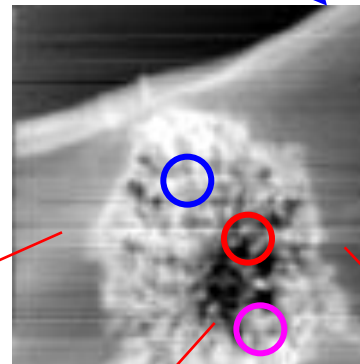
# Problem : distinguish nanoparticles **rutile** - **anatase** (TiO<sub>2</sub>) texture



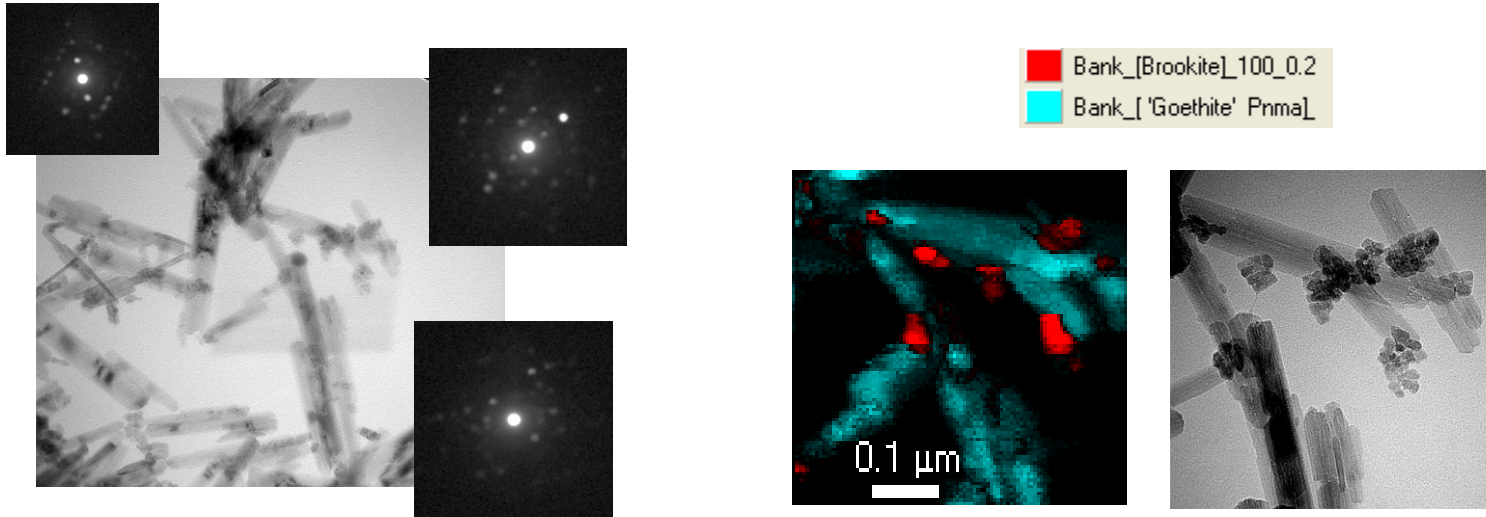
**Rutile**  $P4_2/mnm$   $a = 4.59 \text{ \AA}$ ,  $c = 2.95 \text{ \AA}$

Courtesy Dr. Bakardieva REZ Prague

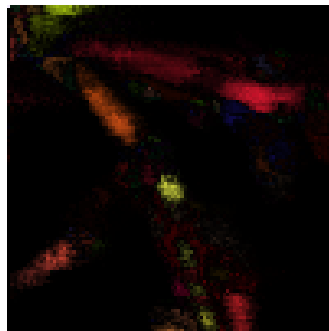
Jeol 3010 – ASTAR



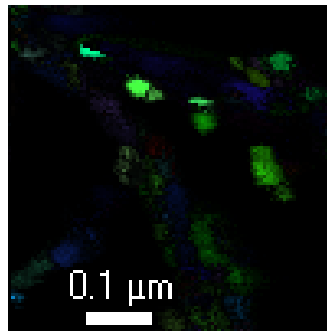
**PROBLEM :** distinguish between **goethite ( $\alpha\text{-FeO(OH)}$ )** 300x40x10nm  
and **brookite ( $\text{TiO}_2$ )** platelets of 48nm diameter



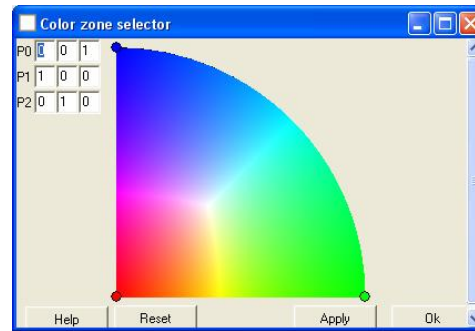
■ Bank\_[Brookite]\_100\_0.2  
■ Bank\_[Goethite] Pnma]



*Orientations ( $\alpha\text{-FeO(OH)}$ )*



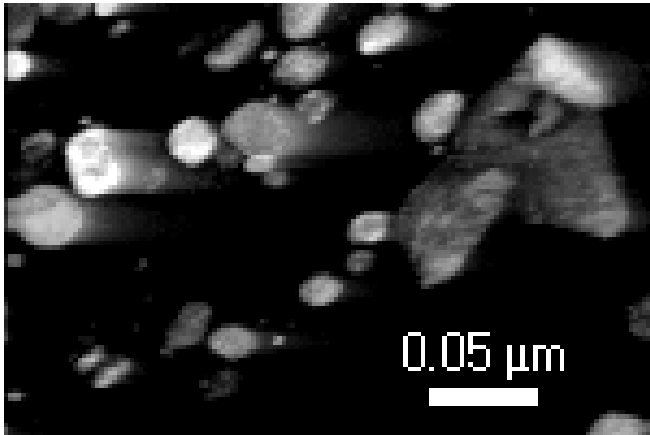
*Brookite  $\text{TiO}_2$*



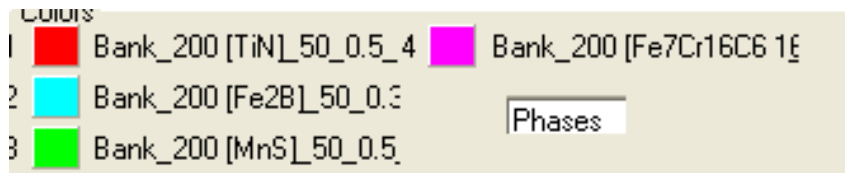
*Code de couleur*

# Libra 200-ASTAR : sub-nanometer (< 1 nm) phase map

Replica with TiN, Fe<sub>2</sub>B, MnS and Fe<sub>6</sub>Cr<sub>6</sub>

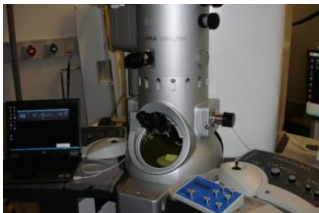
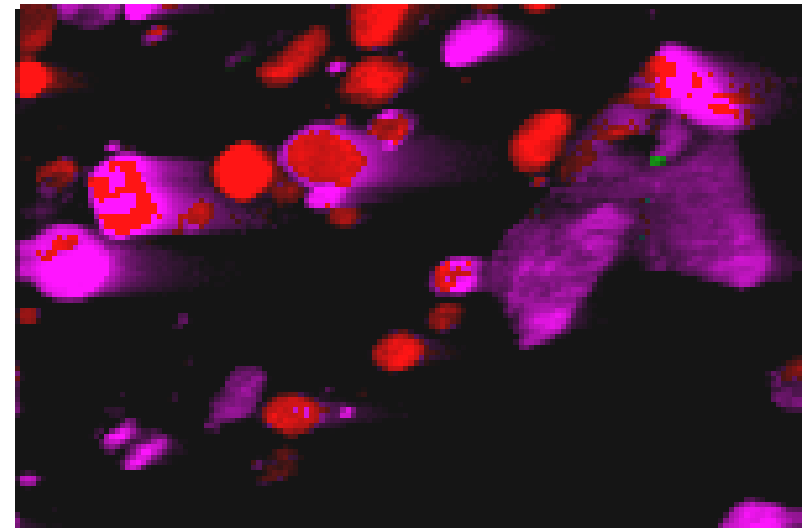


Index correlation map



Phase map can distinguish between 3 phases :

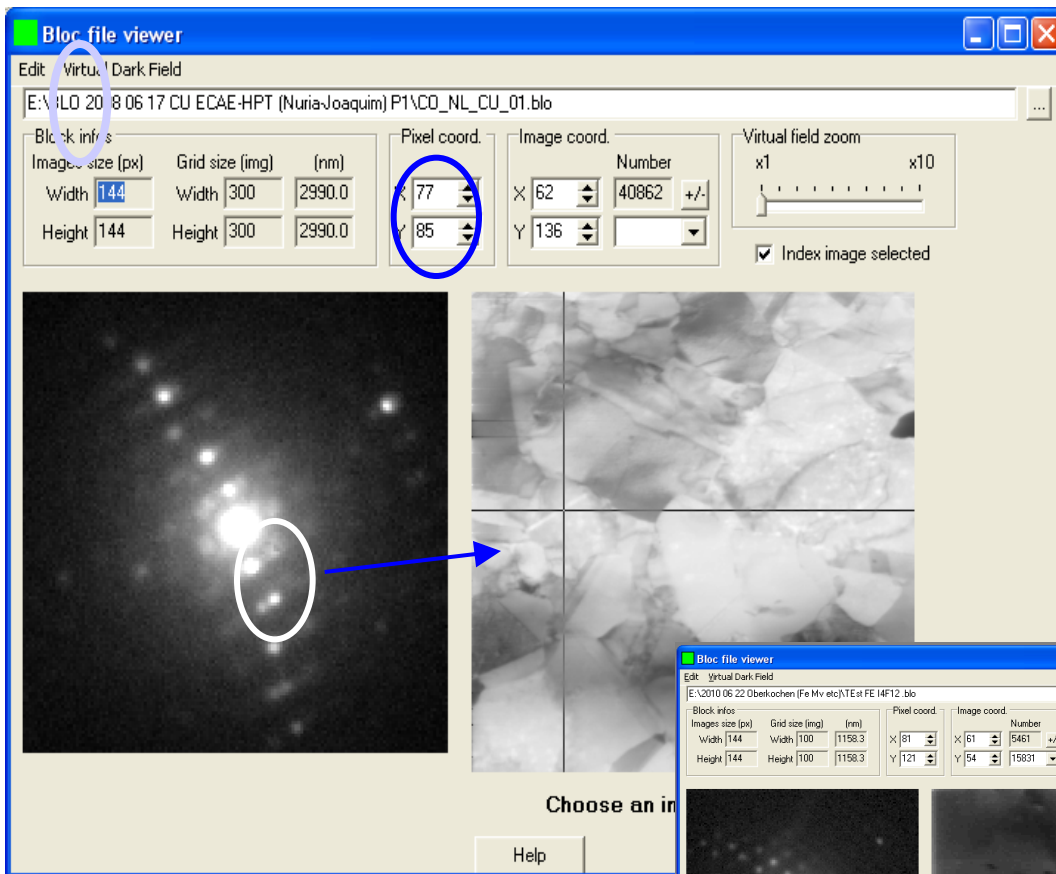
TiN, Fe<sub>2</sub>B, MnS and Fe<sub>6</sub>Cr<sub>6</sub>



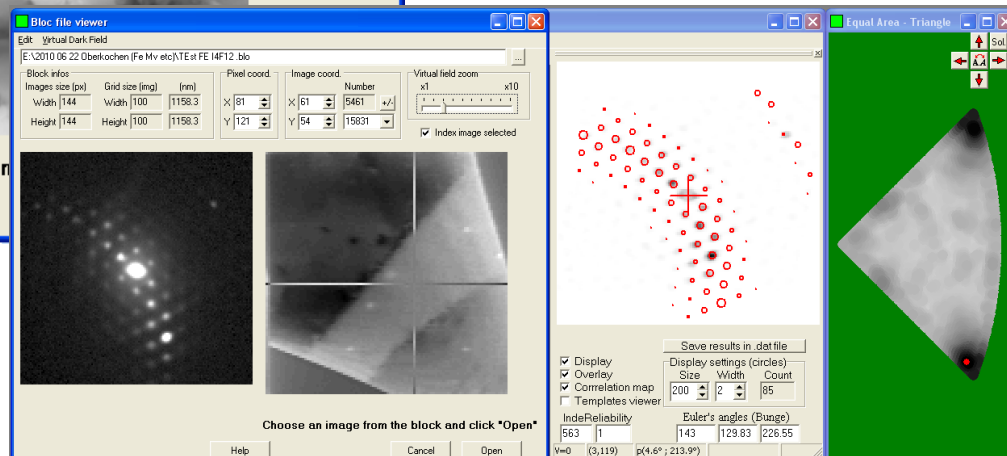
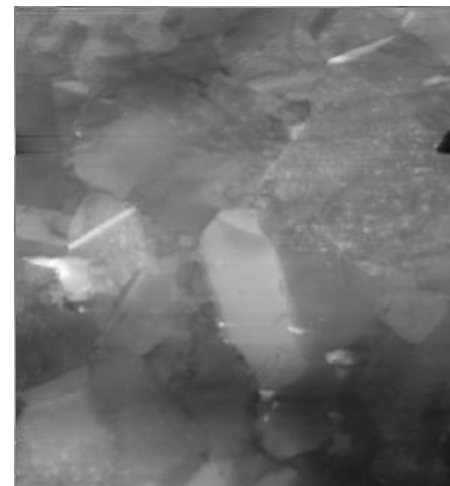
< 1 nm resolution phase map

# INDEX and create virtual dark and bright field maps

Diffraction Pattern viewer with virtual aperure



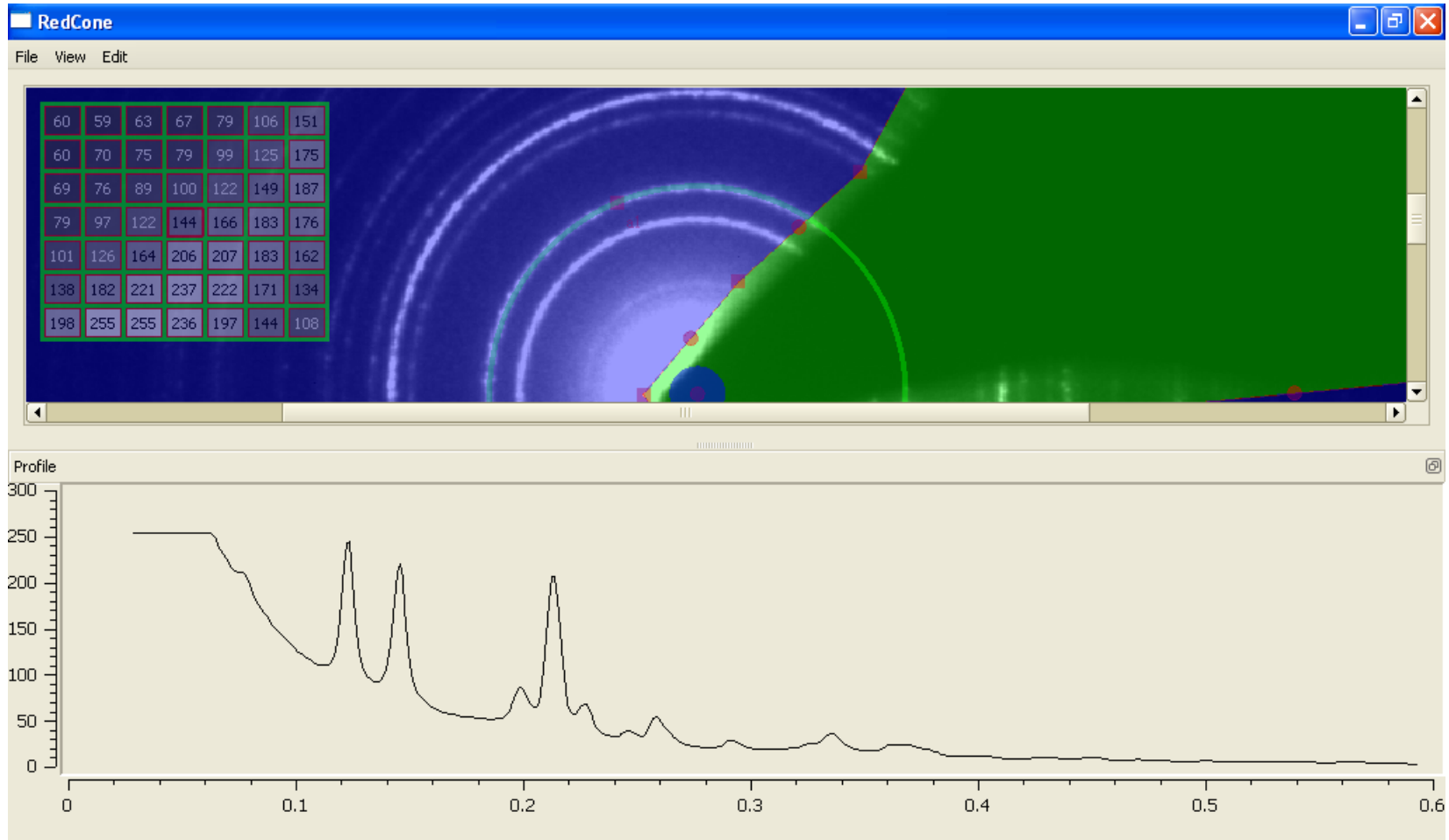
Virtual dark field image





# Polycrystalline Electron diffraction

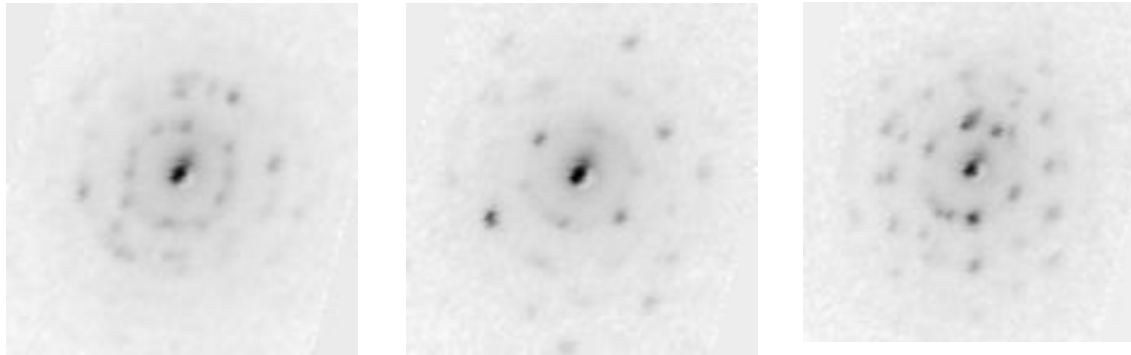
“X-Ray amorphous” OR nanocrystalline ?



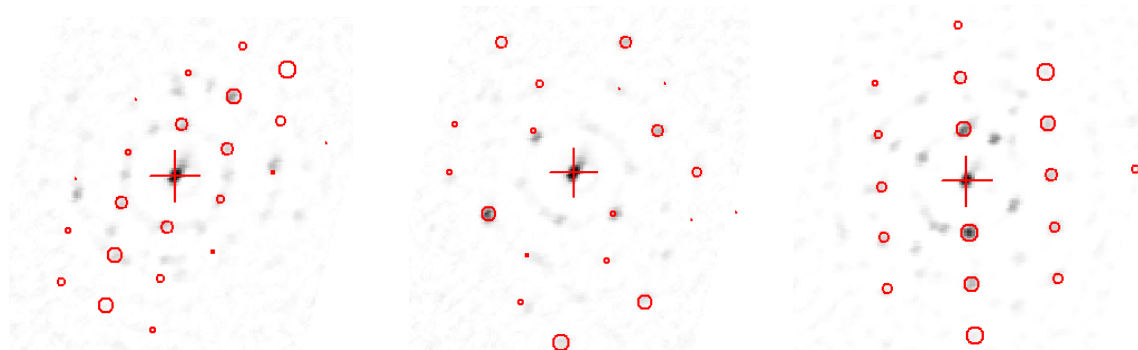
# Orientation mapping on Nanocrystallized polycrystalline Ni-Fe sample

**Polycrystalline thin film of Ni-Fe nanocrystals (average size 5-20 nm )**

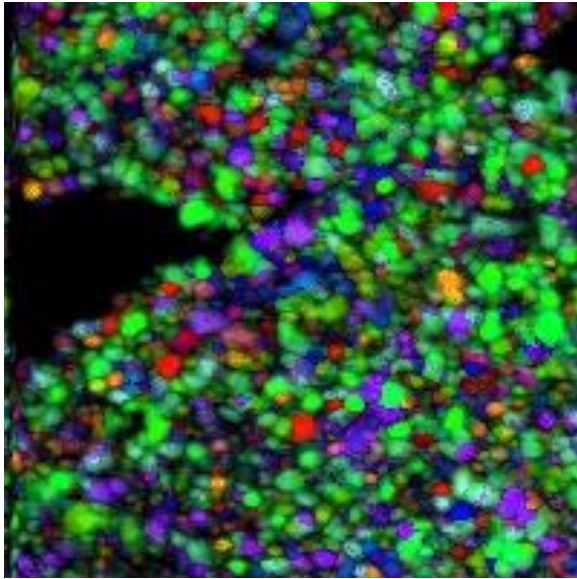
**Data taken with JEOL JEM 2200F operating at 200 kV spot size 1-8 nm**



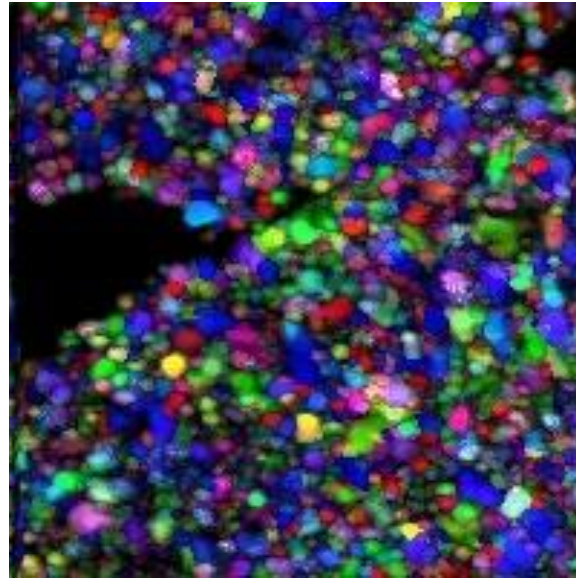
**ASTAR can index even overlapping diffraction patterns from polycrystalline sample**



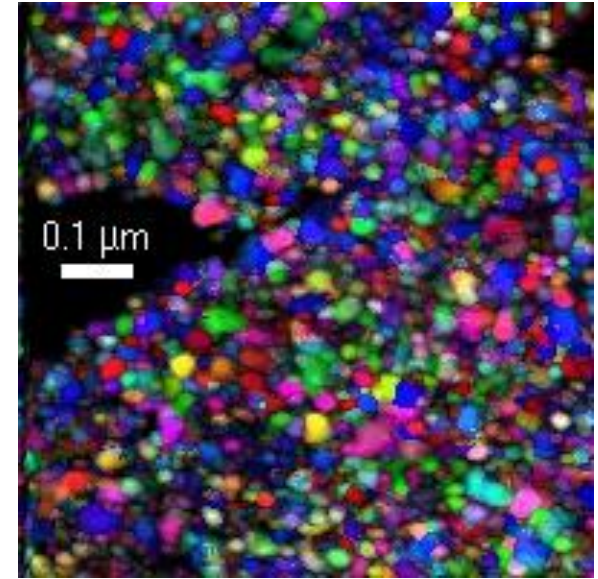
# Crystallographic orientation map



Orientation map (z)  
+ index



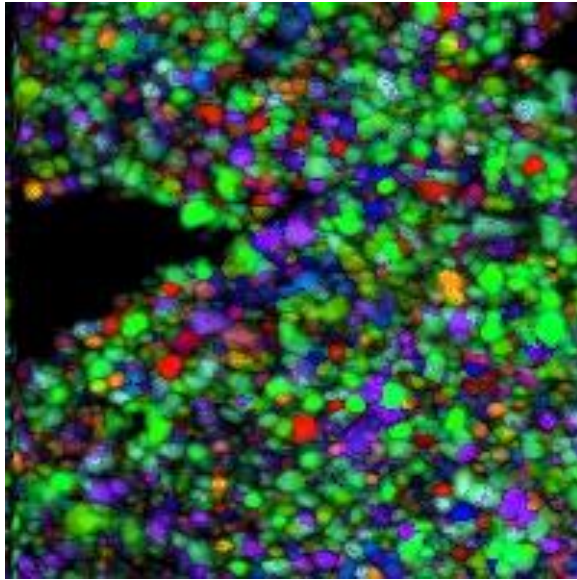
Orientation map (x)  
+ index



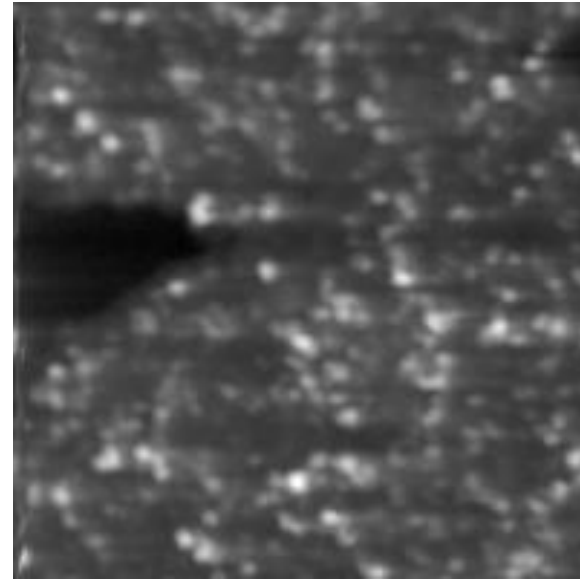
Orientation map (y)  
+ index

As for EBSD equipments, the orientation of one particular axis (usually the z axis) is given through a colour code. This enables the individual crystals to be recognized and/or their size to be estimated (step size 4 nm )

## Step size 4 nm



Orientation map (z)

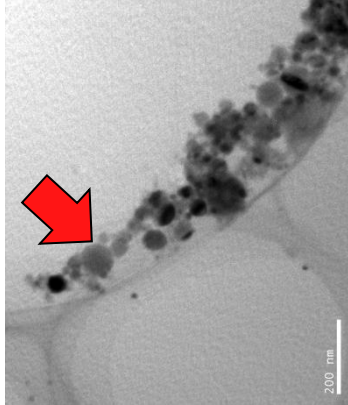


Virtual dark field map

A STEM type image is reconstructed from the collection of diffraction patterns. Virtual Dark Field images (VDF) may be reconstructed as well by selecting a particular reflection placing a virtual aperture on that selected spot on a diffraction pattern

**JEOL 2200 FS 200 KV, Humboldt Univ Berlin**

# Nanoparticle ( 50 nm ) phase identification



cubic 8.32 Å

$Fd\bar{3}m$

**Magnetite or maghemite ??**

$P4_132$   $\gamma\text{-Fe}_2\text{O}_3$

$\text{Fe}_3\text{O}_4$

cubic 8.32 Å



Orientation map precession  $0.3^\circ$



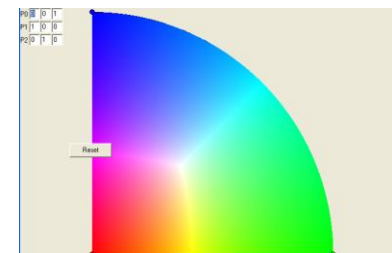
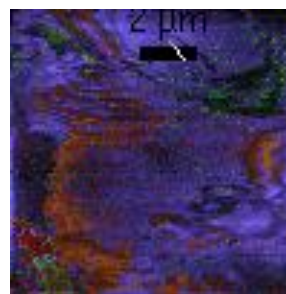
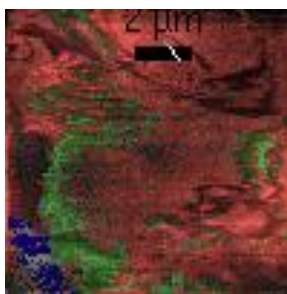
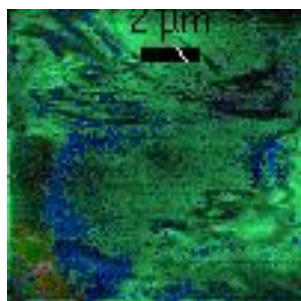
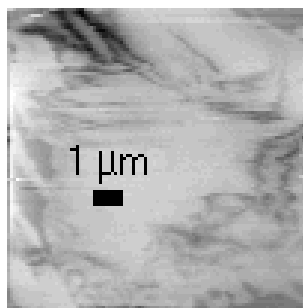
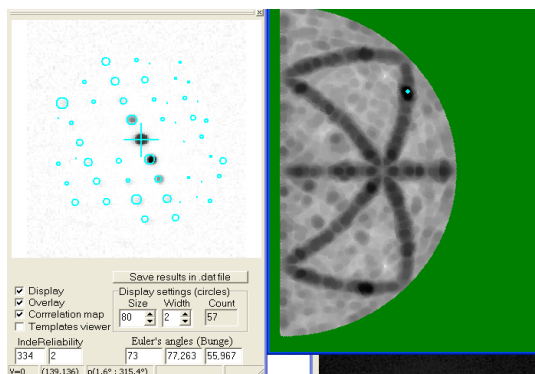
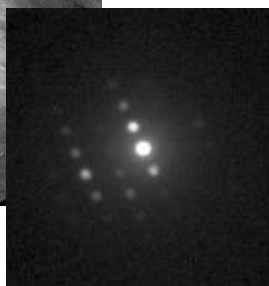
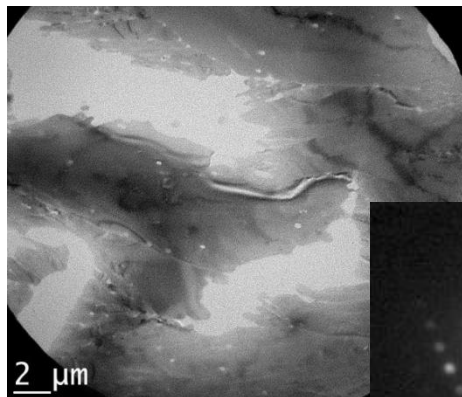
PHASE map precession  $0.3^\circ$

ALL Nanoparticles  
REVEALED AS  
**magnetite (RED )**

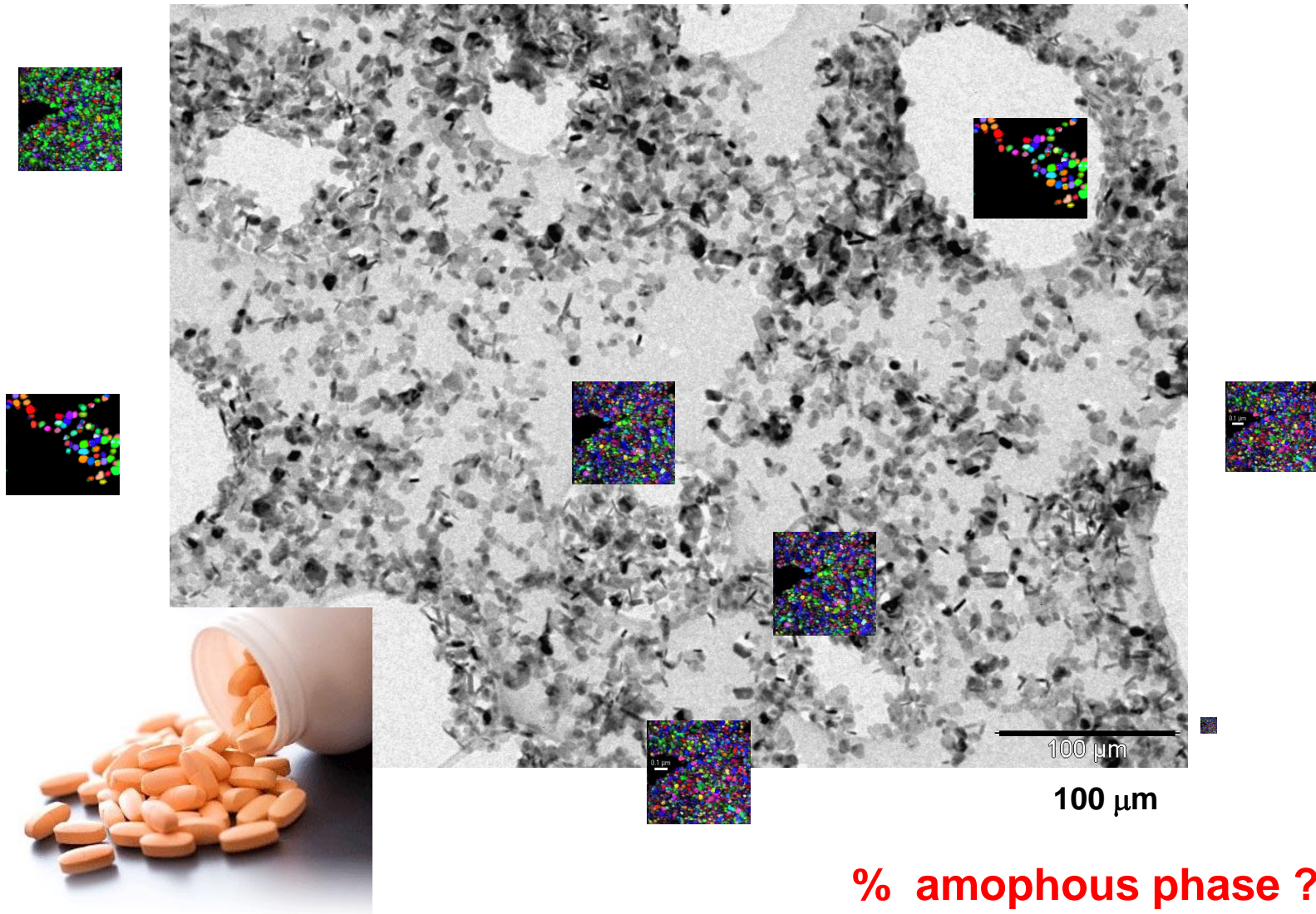
# New application : ASTAR on organic structures

TRIS structure  $C_{16}H_{48}N_4O_{12}$

Pna2<sub>1</sub> cell 0.7768 X 0.8725 X 0.8855 nm



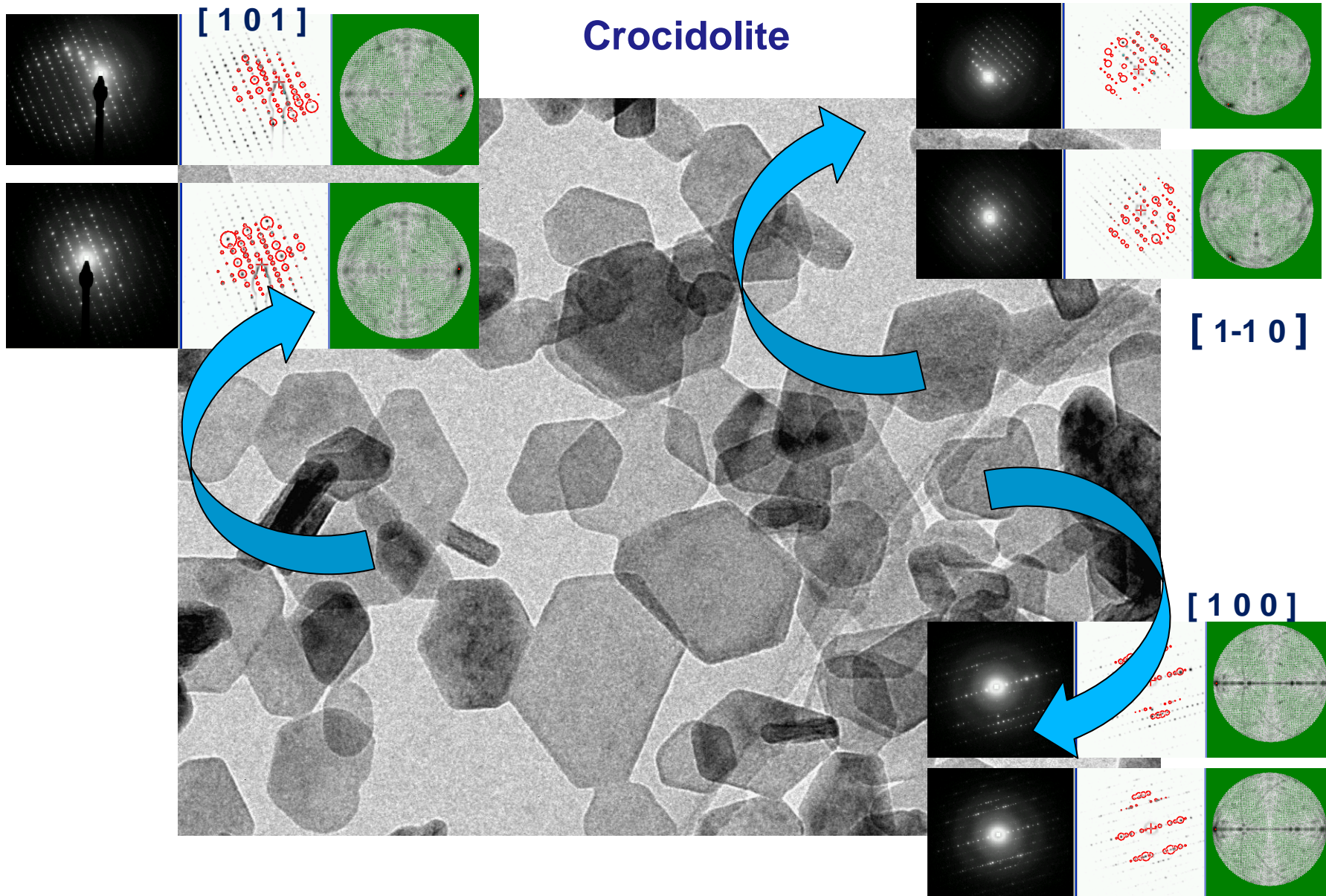
# ASTAR texture – (multi) phase analysis from different sample areas



What is the % of each phase ?

% amorphous phase ?

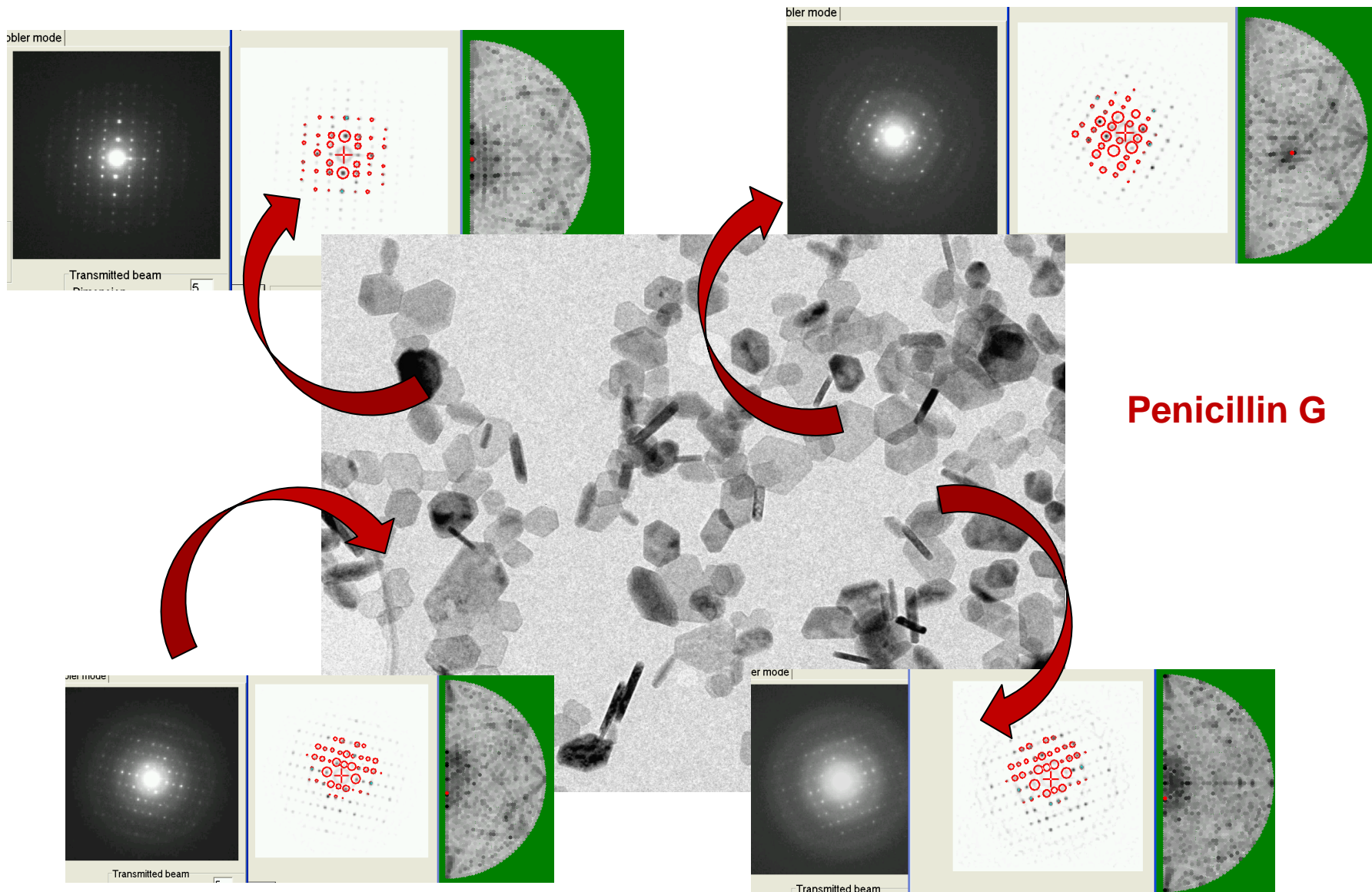
detection limit ?



“Random 3D tomography” - ASTAR

Cell determination & basic structure model building - **INORGANIC CRYSTALS**

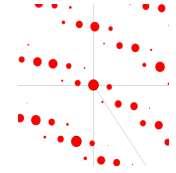
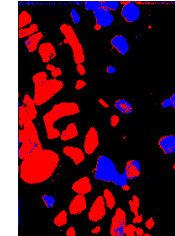
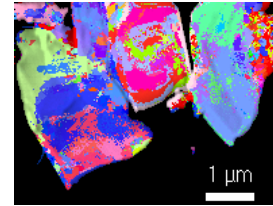
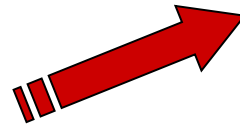
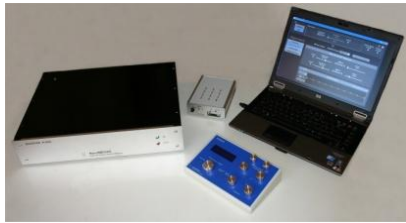




**“Random 3D tomography” - ASTAR**

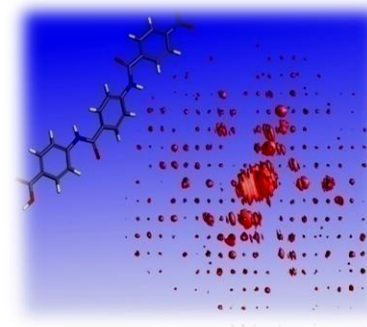
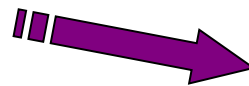
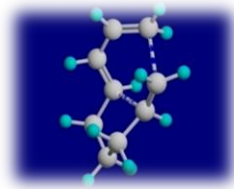
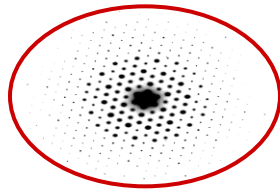
**Cell determination & basic structure model building – ORGANIC CRYSTALS**

# TEM advanced ELECTRON DIFFRACTION SOLUTIONS for pharma-organic crystals

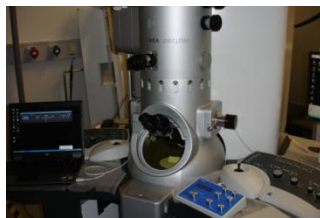


AUTOMATIC ORIENTATION / PHASE MAPPING

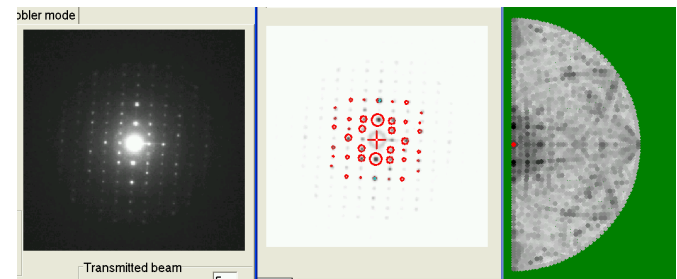
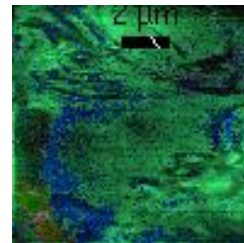
**ASTAR**



**ADT -3D diffraction tomography**



**PRECESSION**  
« SpinningStar- DigiSTAR »



# **ASTAR combinations (examples)**

**Laurence Livermore Lab (Berkley –California) USA  
CM300 FEG – Jeol 2000**

**Portland State Univ –Oregon USA  
Tecnai 20F - Jeol 2000**

**Alabama Univ USA Tecnai 30F**

**Univ of Texas –Austin USA Jeol 2010 F**

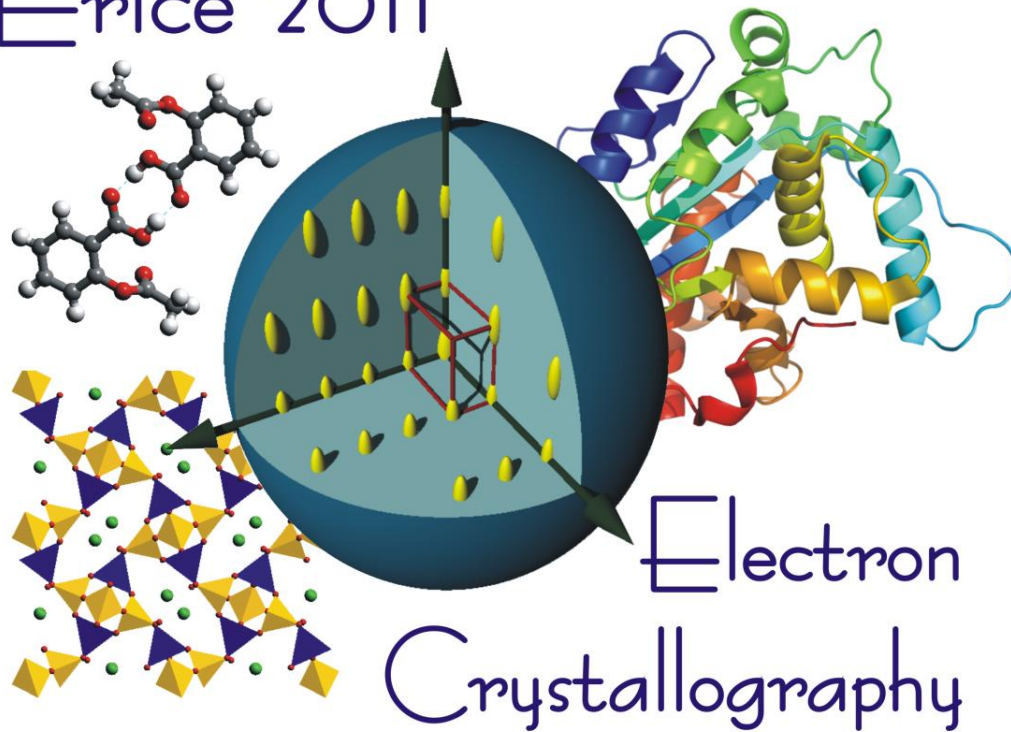
**Humboldt Univ Berlin Jeol 2200 FS**

**KBSI Korea Jeol 2100F**

**Osaka Gas , Japan Jeol 2100F**

**Zeiss Application Lab (Germany) Libra 200 FE Cs – Libra 120**

Erice 2011



**NATO summer school in Erice, Sicily  
2.-12. June 2011 (parallel to Powder Diffraction)  
[www.crystalerice.org](http://www.crystalerice.org)**

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*Thanks for your attention !!*